Methoden moderner Röntgenphysik II: Streuung und Abbildung

Lecture 11	Vorlesung zum Haupt- oder Masterstudiengang Physik, SoSe 2018 G. Grübel, A. Philippi-Kobs, O. Seeck, T. Schneider, L. Frenzel, M. Martins, W. Wurth		
Location	Lecture hall AP, Physics, Jungiusstraße		
Date	Tuesday	13:00 - 14:30	(starting 3.4.)
	Thursday	8:30 - 10:00	(until 12.7.)

Outline

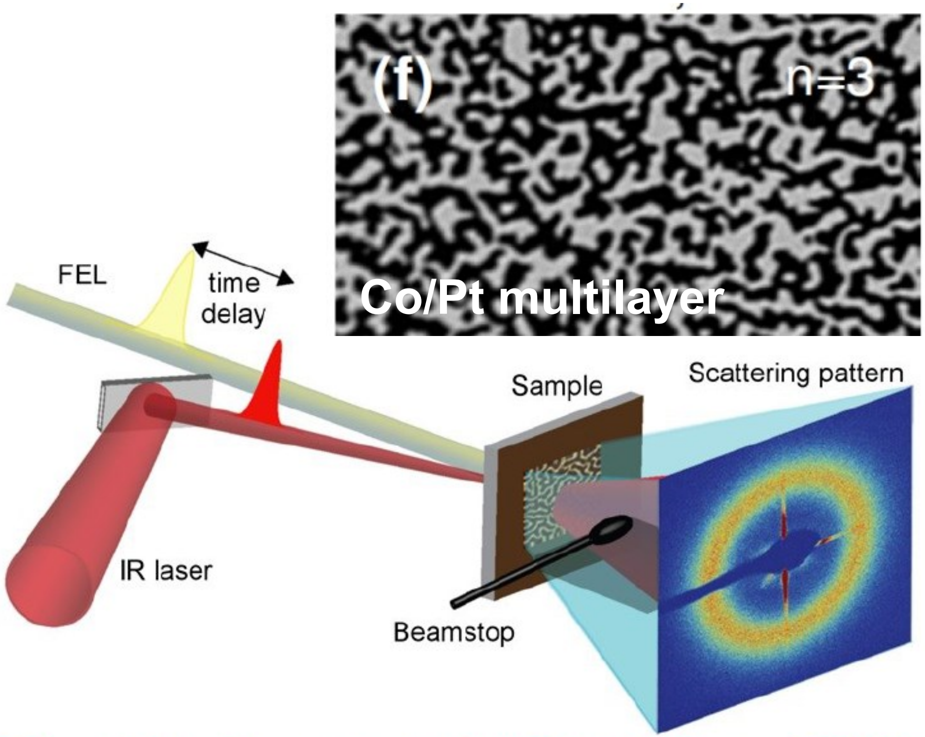
Part II/1:

Studies on Magnetic Nanostructures

by André Philippi-Kobs (AP)

[15.5.] **Ferromagnetism in a Nutshell**

- Introduction to Magnetic Materials
- Magnetic Phenomena
- Magnetic Free Energy
- Perpendicular Magnetic Anisotropy
- Magnetic Domains and Domain Walls



B. Pfau et al., Nature Communications, Vol. 3, 11; DOI:doi:10.1038/ncomms2108 (2012)
L.Müller et al., Rev. Sci. Instrum. 84, 013906 (2013)

[17.5.] **Interaction of Polarized Photons with Ferromagnetic Materials**

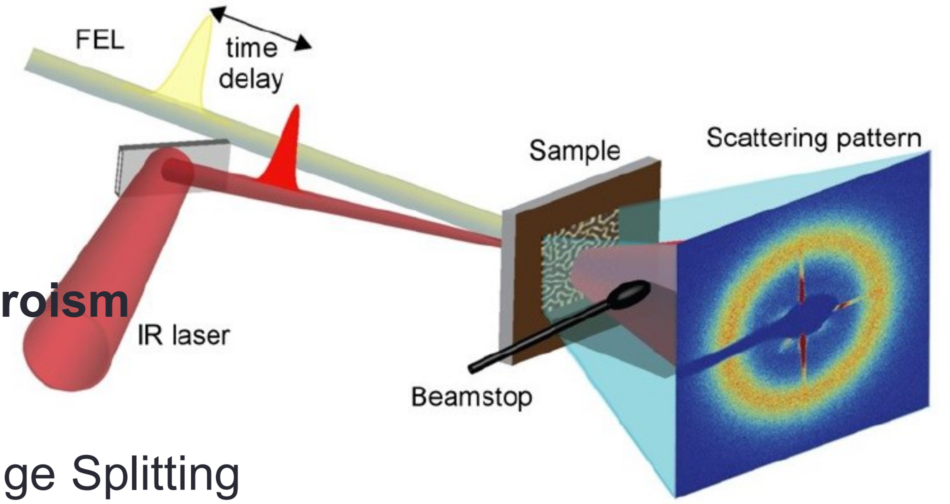
- Charge and Spin X-ray Scattering by a Single Electron
- Absorption and Resonant Scattering of Ferromagnets (Semi-Classical and Quantum-Mechanical Concepts)

Outline

Part II/2:

Studies on Magnetic Nanostructures

by André Philippi-Kobs (AP)

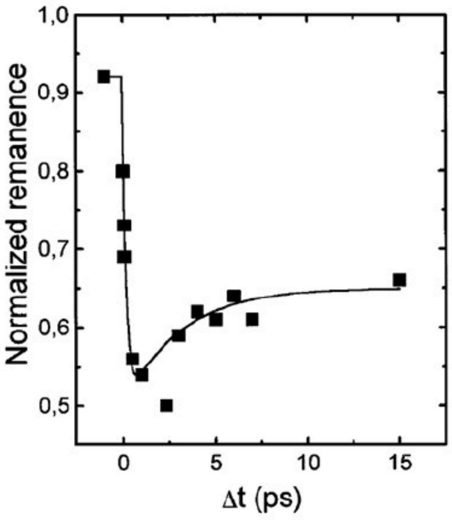


[~~29.5.~~19.6.] X-ray Magnetic Circular Dichroism (XMCD) & Resonant Magnetic Small Angle X-ray Scattering (mSAXS)

- Role of Spin-Orbit Coupling and Exchange Splitting
- Sum Rules
- XMLD and Natural Dicroisms
- mSAXS of Magnetic Domain Patterns

[~~31.5~~.**21.06.**] Femtomagnetism

- Introduction to Ultrafast Magnetization Dynamics Induced by Femtosecond Infrared Pulses
- Pump-Probe Experiments of Nano-Scale Magnetic Domain Patterns
- All-Optical Switching
- Manipulating Magnetism by XUV and THz Pulses



Outline

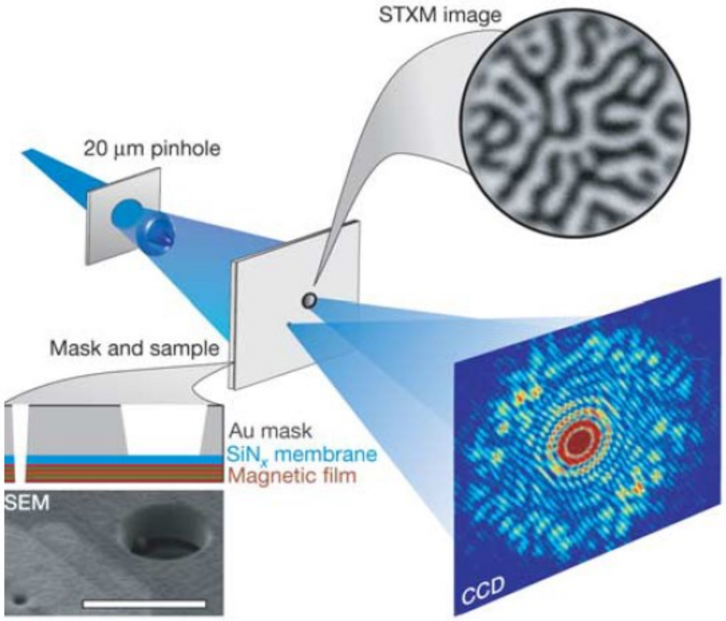
Part II/3:

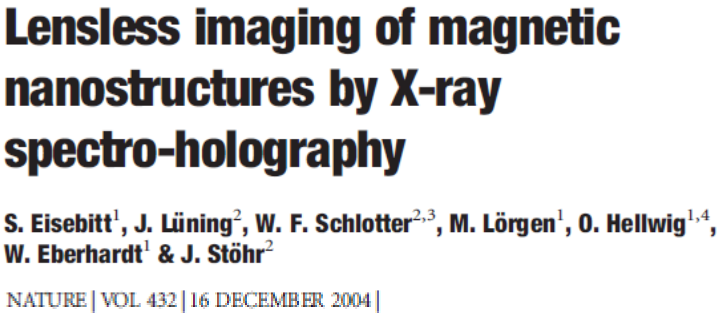
Studies on Magnetic Nanostructures

by André Philippi-Kobs (AP)

[26.6.] Imaging of Magnetic Domains

- **Fourier Transform Holography (FTH)**
- Scanning Transmission X-ray Microscopy (STXM)
- Coherent Diffraction Imaging (CDI)

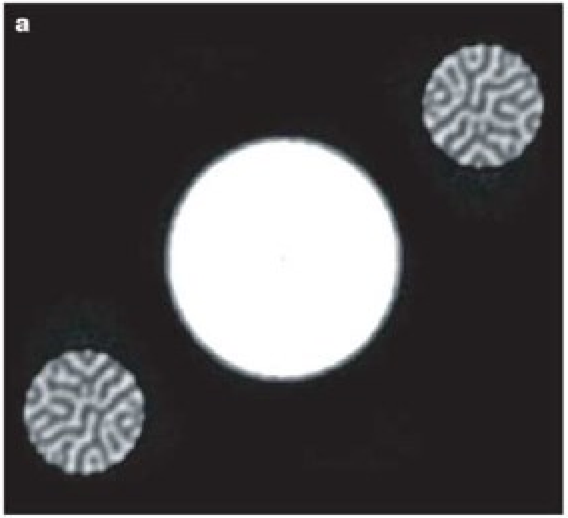




Lensless imaging of magnetic nanostructures by X-ray spectro-holography

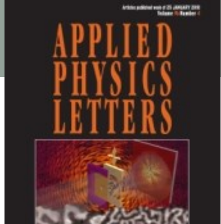
S. Eisebitt[1], J. Lüning[2], W. F. Schlotter[2,3], M. Lörgen[1], O. Hellwig[1,4], W. Eberhardt[1] & J. Stöhr[2]

NATURE | VOL 432 | 16 DECEMBER 2004 |



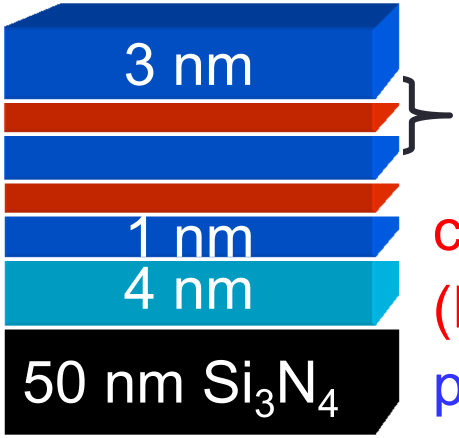
FTH

Co/Pt Multilayers



Layer Composition

multilayer stack (n = 1 - 32)



x (n - 1)

cobalt
(DC magnetron)
platinum
(DC magnetron)
platinum (ECR)

Perpendicular magnetic anisotropy in Co/Pt discovered in 1988

Carcia et al., J. Appl. Phys. **63**, 5066 (1988)

Magnetic Domains

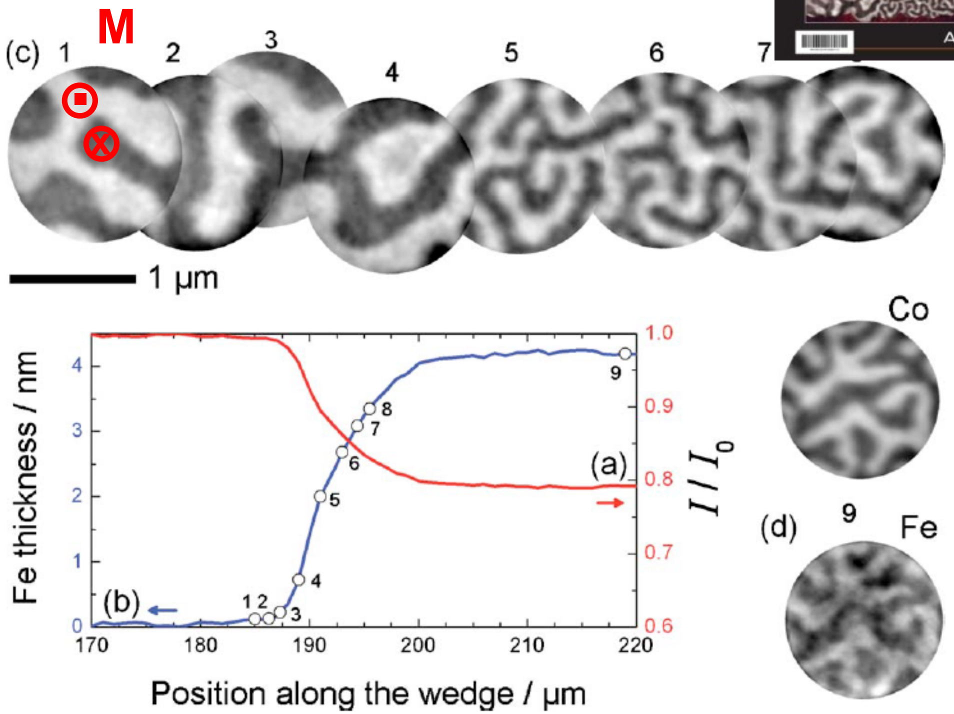


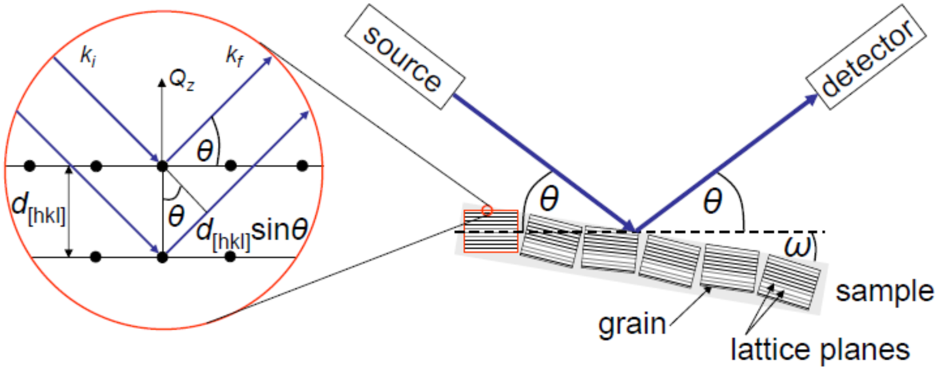
FIG. 3. (Color online) Domain size evolution of a Co/Pt multilayer film covered by an iron wedge. Plot (a) gives the absorption profile (normalized photodiode current) at the Fe L_3 absorption edge when scanning over the Fe wedge. The absorption is used to calculate the local iron overlayer thickness (b). A contiguous series of XMCD holograms at the Co L_3 absorption edge has been acquired and reconstructed (c) at the indexed positions along the iron wedge. Image (d) has been measured at the Fe L_3 absorption edge at the very same position as the last Co image (#9).

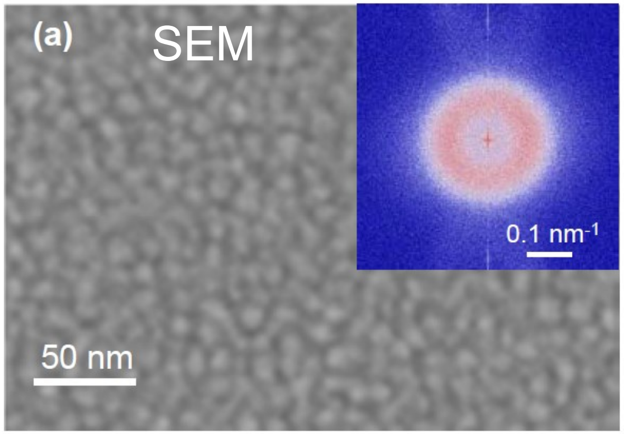
Stickler, G. Grübel, H. P. Oepen et al., Appl. Phys. Lett. **96**, 042501 (2010)

Co/Pt Multilayers

Structural analysis

X-ray diffraction (XRD) to determine crystallinity





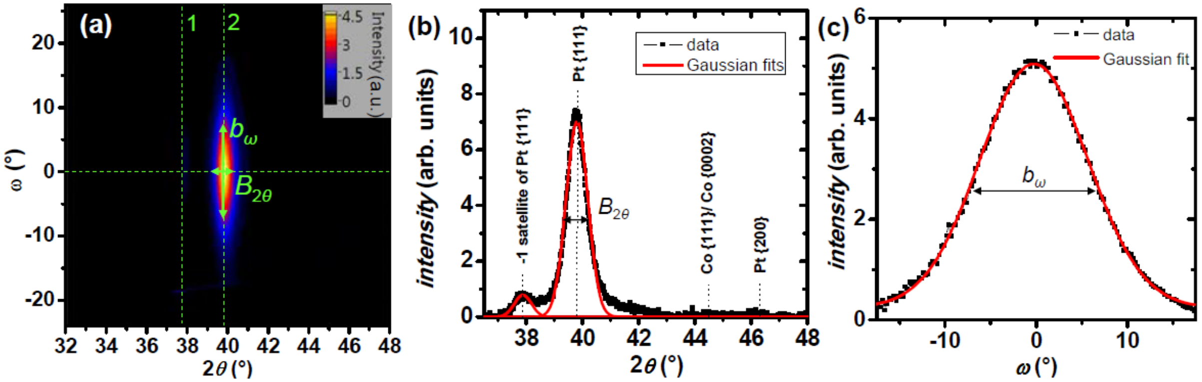


Figure 5.17: (a) Diffraction map $I(\omega, 2\theta)$ of a 5 nm Pt/ (0.8 nm Co/ 4 nm Pt)$_3$/ 0.8 nm Co/ 3 nm Pt multilayer grown on Si_3N_4. The intensity is color coded according to the given color bar. The positions of the -1 satellite reflex (1) and of the peak at the Pt(111) position (2) are indicated by vertical dashed lines. (b) shows the integrated intensity $I(2\theta) = \sum_\omega I(\omega, 2\theta)$ and (c) the cross-section $I(\omega)$ at the peak position $2\theta_{\text{fcc Pt(111)}} = 39.8°$. Both curves are fitted to a normal distribution (red lines) with a FWHM of $B_{2\theta}$ and b_ω, respectively.

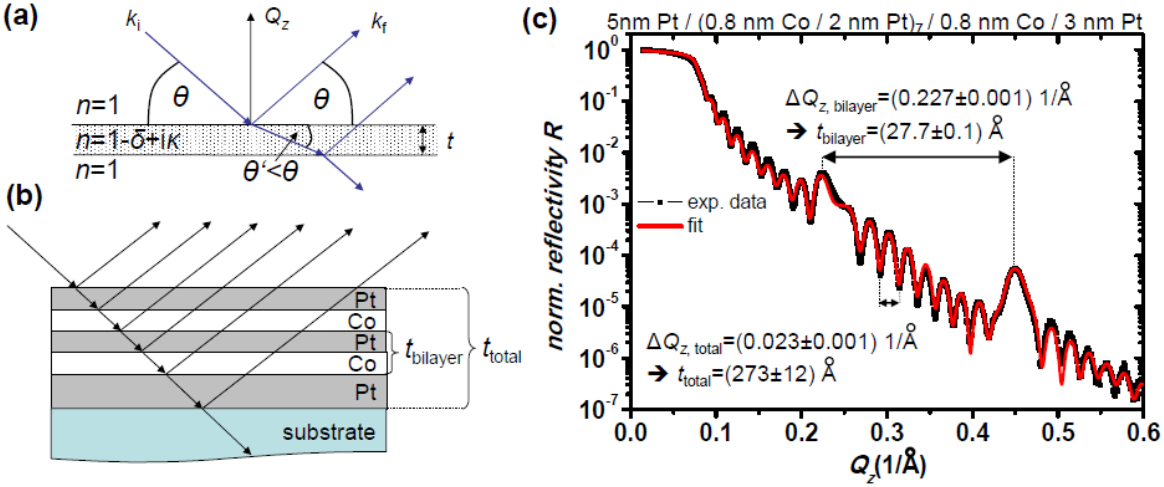
- Co and Pt layers have fcc lattice
- polycrystalline,
  grain size of (11 ± 2) nm
- out-of-plane textured,
  tilting of grains (FWHM):(23 ± 2)°

More details, see G. Winkler, A. Kobs, A. Chuvilin, D. Lott, A. Schreyer, H. P. Oepen, J. Appl. Phys. **117**, 105306 (2015)

Co/Pt Multilayers

Structural analysis

X-ray reflectometry (XRR) to determine quality of layered structure



$$\Delta Q_{z,\text{ total}} = \frac{2\pi}{t_{\text{total}}}$$

$$\Delta Q_{z,\text{ bilayer}} = \frac{2\pi}{t_{\text{bilayer}}}$$

- roughness: (0.2 ± 0.1) nm
- interdiffusion of Co & Pt: (0.5 ± 0.2) nm

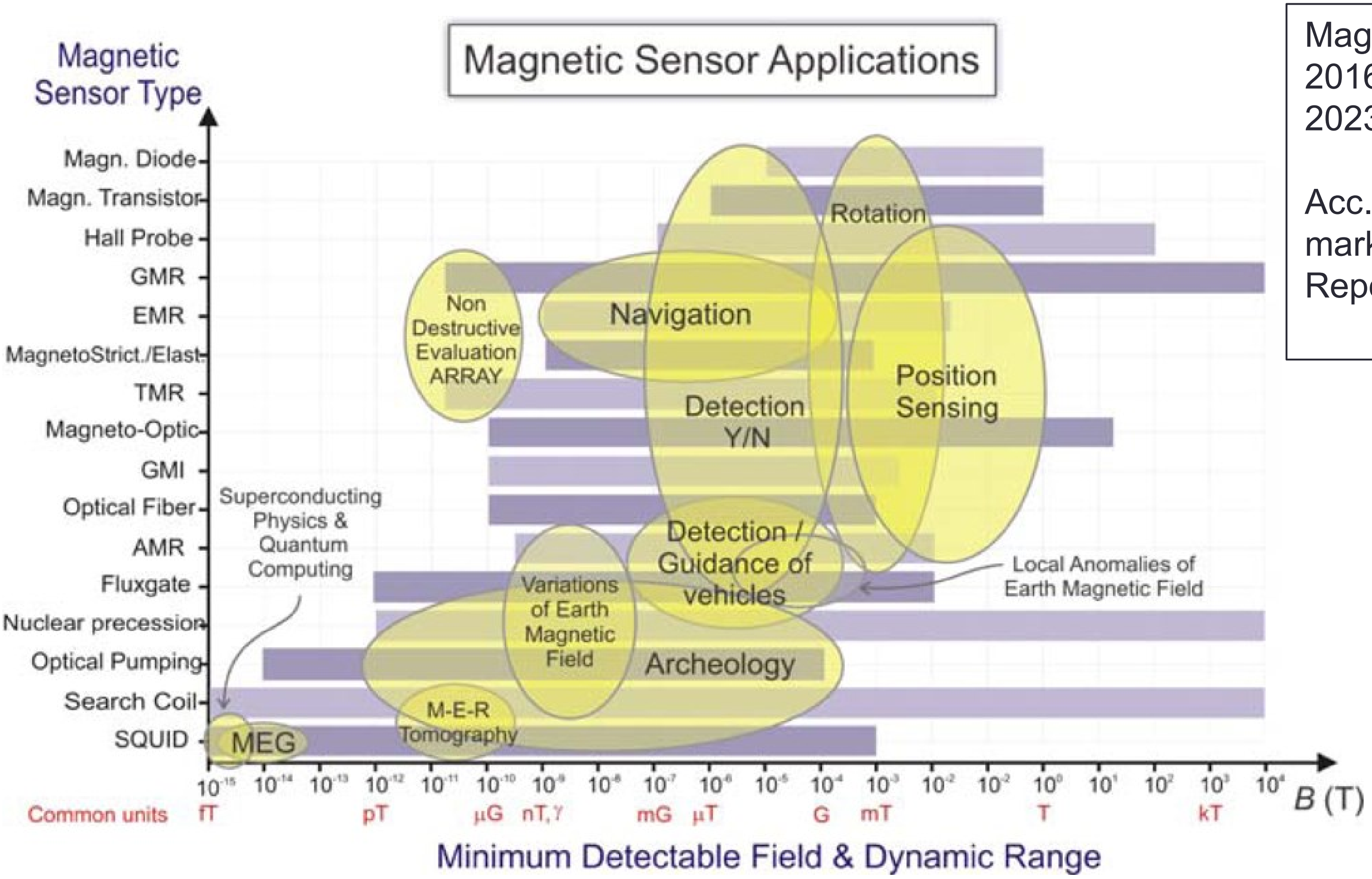
disentangling of both from off-specular scans

Figure 5.27: (a) Refraction and reflection of an x-ray beam hitting a thin layer with thickness t. The interference of the partial waves refracted from the two interfaces generates oscillations (Kiessig fringes) in the reflectivity profile $R(\theta)$. (b) In a periodically layered structure the interferences of the reflected partial waves additionally yield beating waves in $R(\theta)$. (c) Reflectivity R in dependence of the scattering vector Q_z for a multilayer with $n = 8$ and a Pt interlayer thickness of $t_{\text{Pt}} = 2$ nm. From the oscillation and beating wave period the total thickness of the stacking and the bilayer thickness was verified utilizing Eq. 5.64 and Eq. 5.65, respectively. The red solid line is a fit utilizing the software PARRAT32 [715], which is used in particular to determine the thickness of the roughness/interdiffusion regions.

A. Kobs, PhD thesis, Universität Hamburg (2013)

Introduction to Magnetic Materials

Magnetic Materials in Sensor Applications



M. Díaz-Michelena, Sensors 9, 2271 (2009)

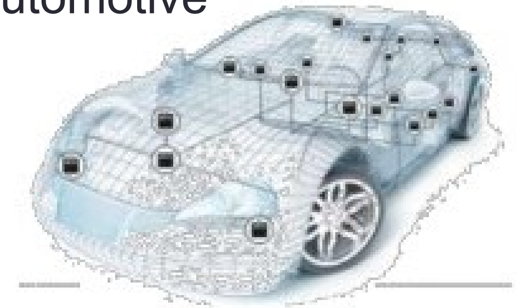
Magnetic Sensor Market:
2016: USD 2.96 Billion
2023: USD 5.37 Billion

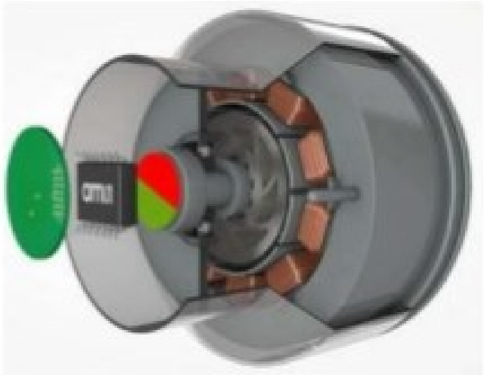
Acc. to marketsansmarkets.com, Report Code: SE 2688

Introduction to Magnetic Materials

Magnetic Materials in Sensor Applications

Automotive

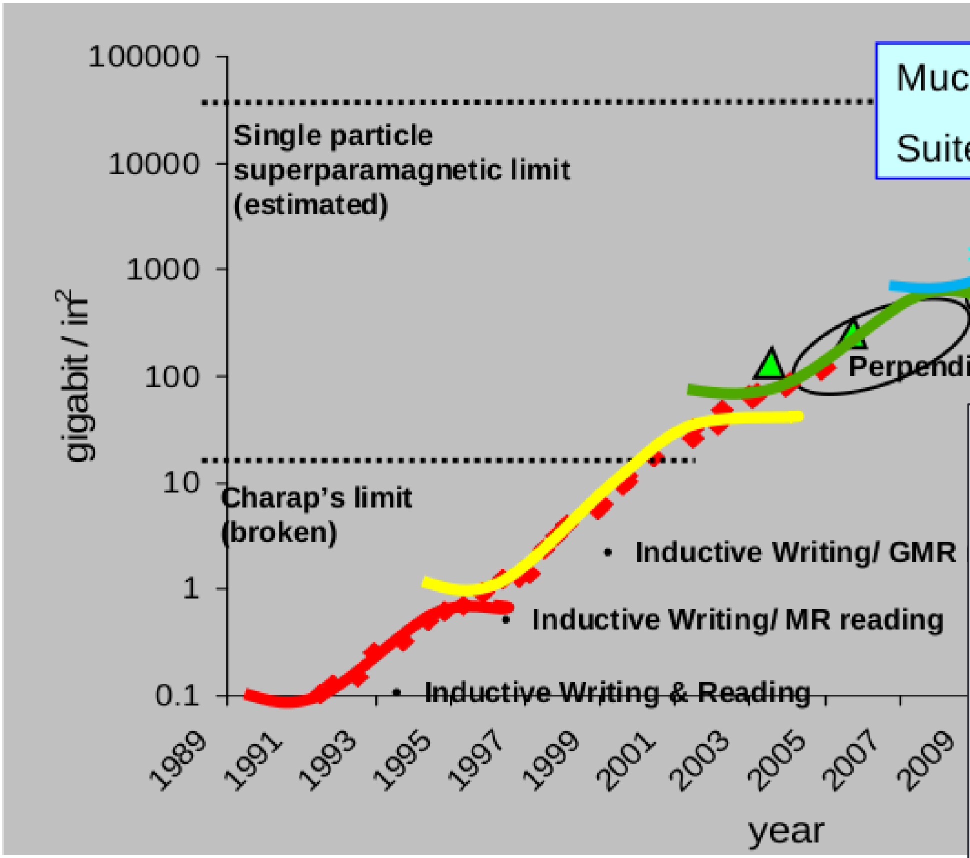




https://ams.com/

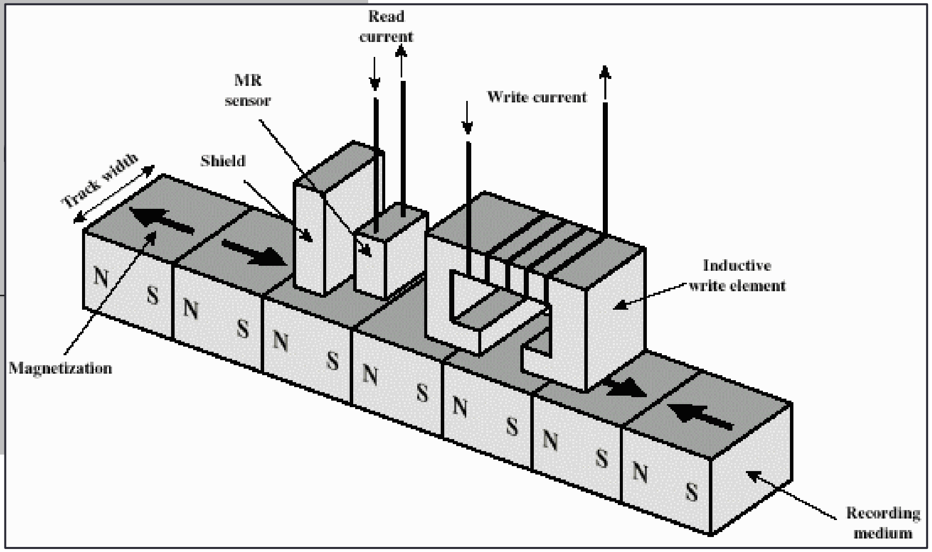
Introduction to Magnetic Materials

Temporal Evolution of Storage Density in HDD



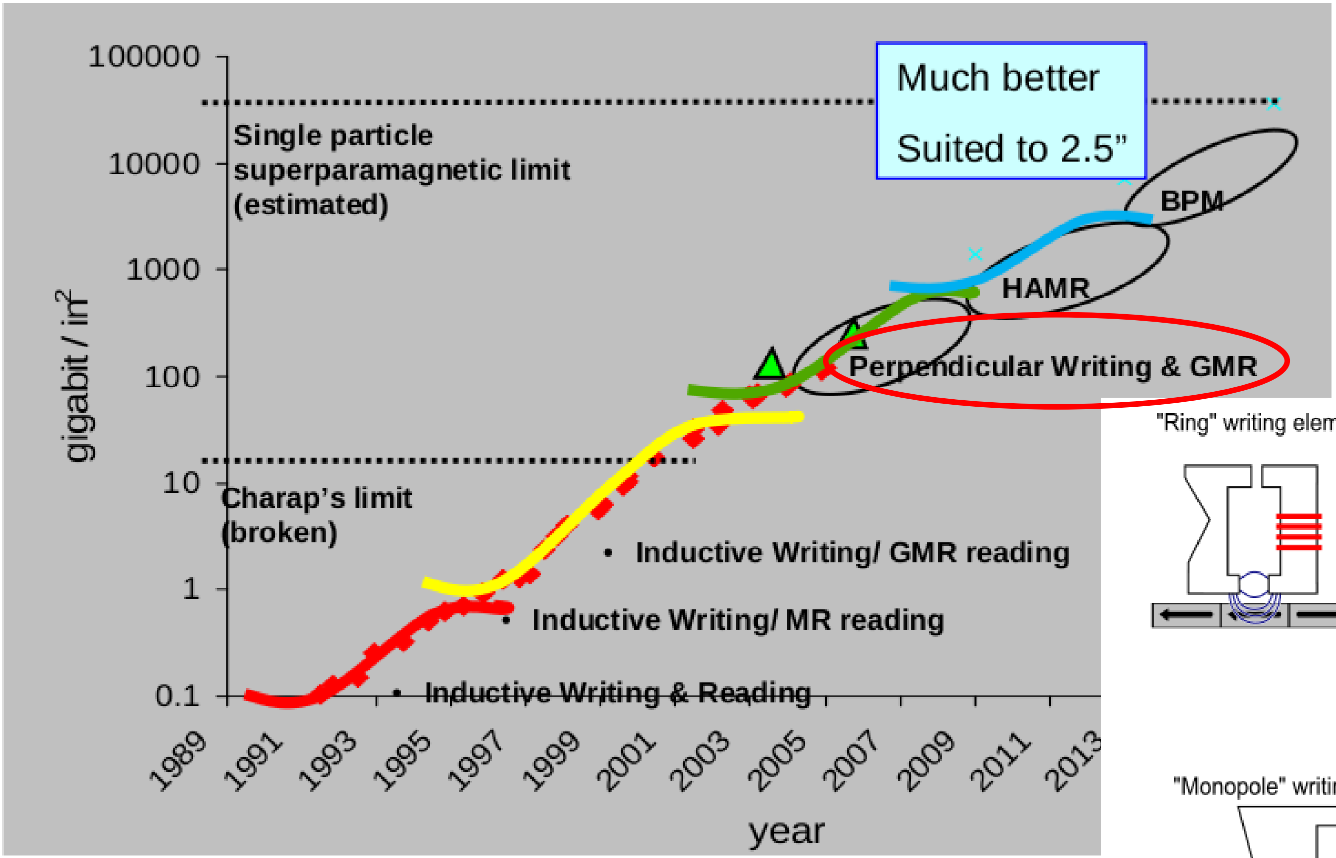
Much better

Suited to 2.5"

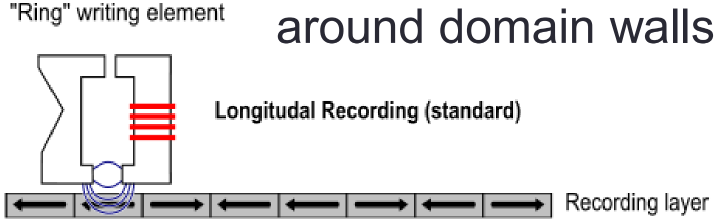


Introduction to Magnetic Materials

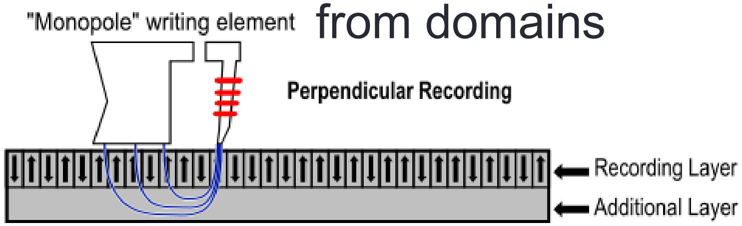
Temporal Evolution of Storage Density in HDD



Low stray field around domain walls

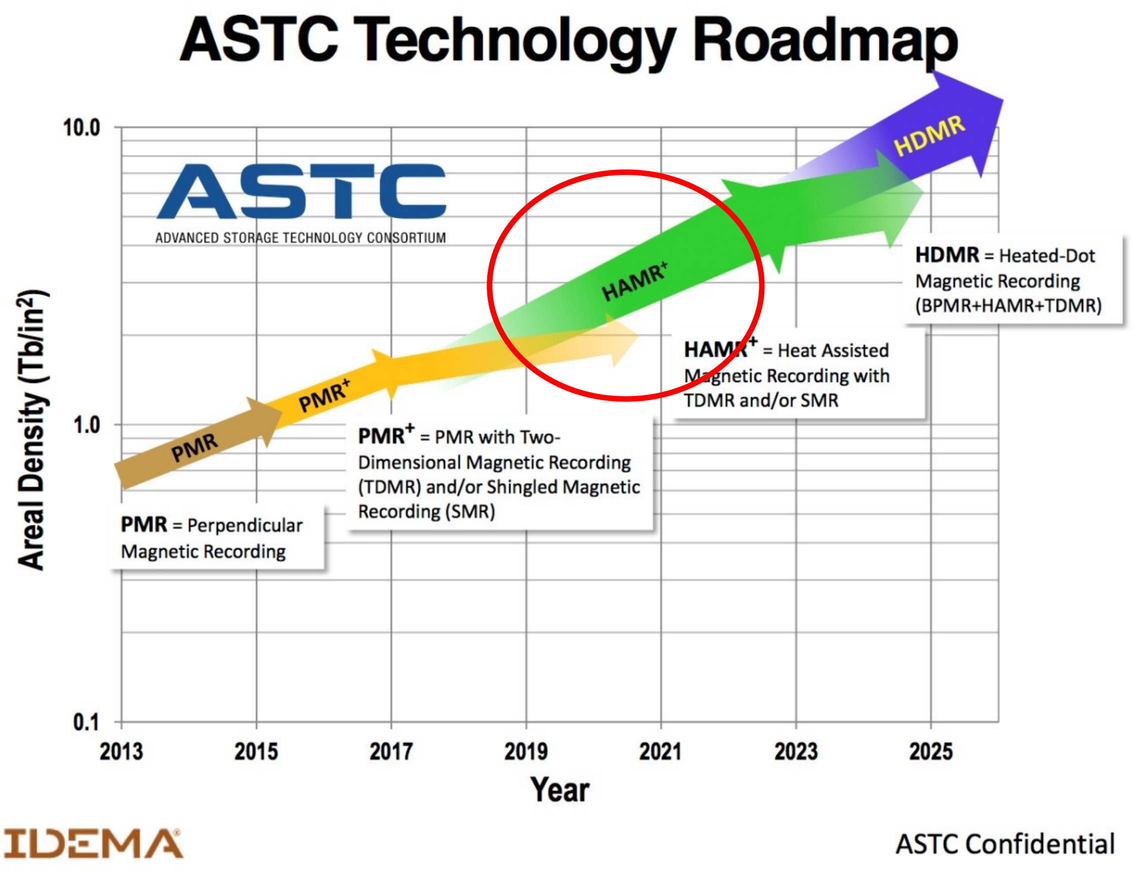


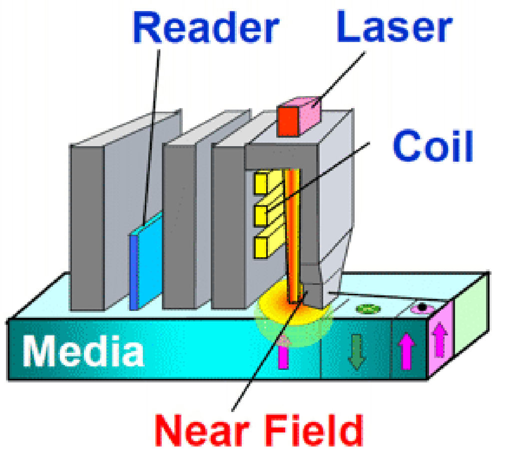
Huge stray field from domains

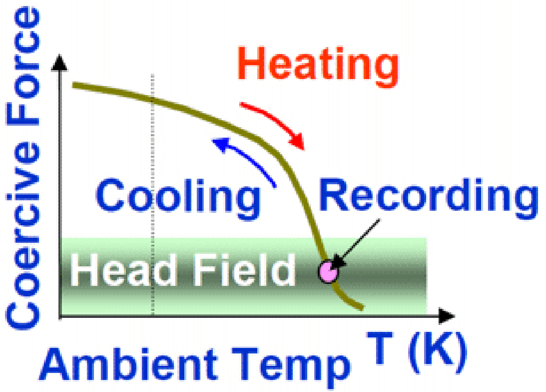


Introduction to Magnetic Materials

Temporal Evolution of Storage Density in HDD

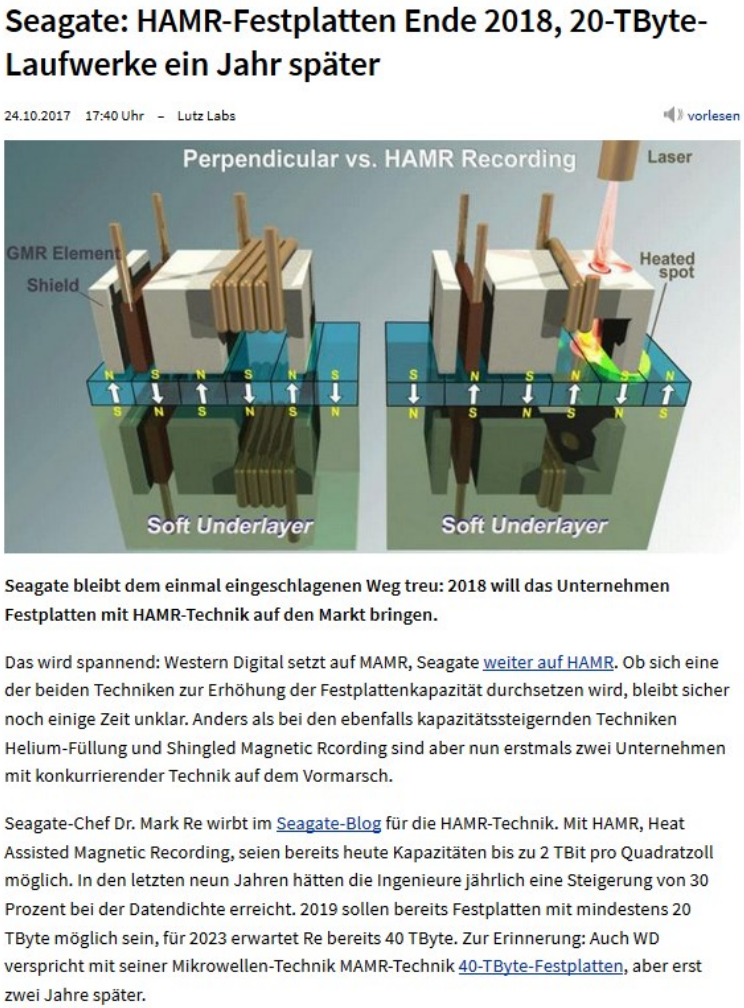






Introduction to Magnetic Materials

Temporal Evolution of Storage Density in HDD



Seagate: HAMR-Festplatten Ende 2018, 20-TByte-Laufwerke ein Jahr später

24.10.2017 17:40 Uhr – Lutz Labs

vorlesen

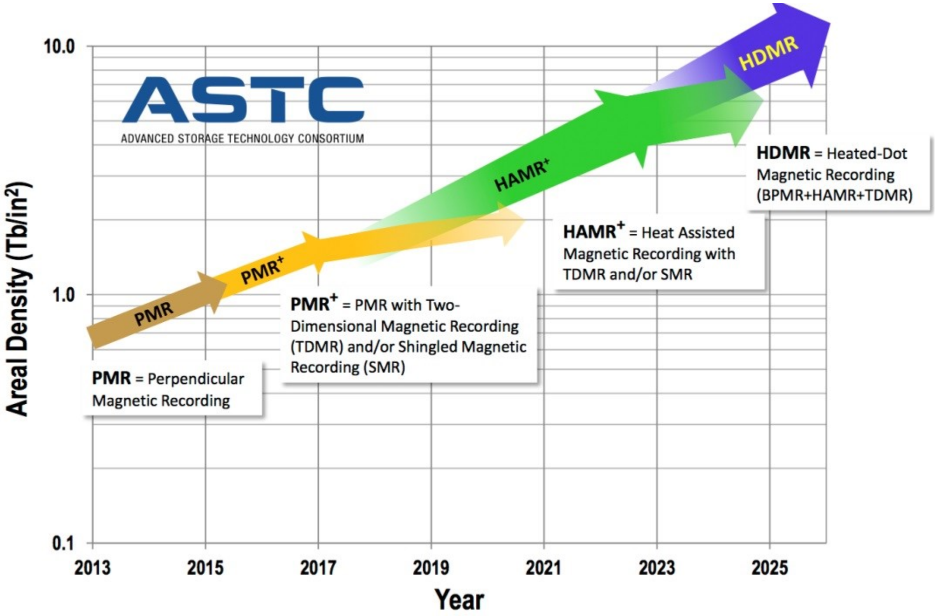
Seagate bleibt dem einmal eingeschlagenen Weg treu: 2018 will das Unternehmen Festplatten mit HAMR-Technik auf den Markt bringen.

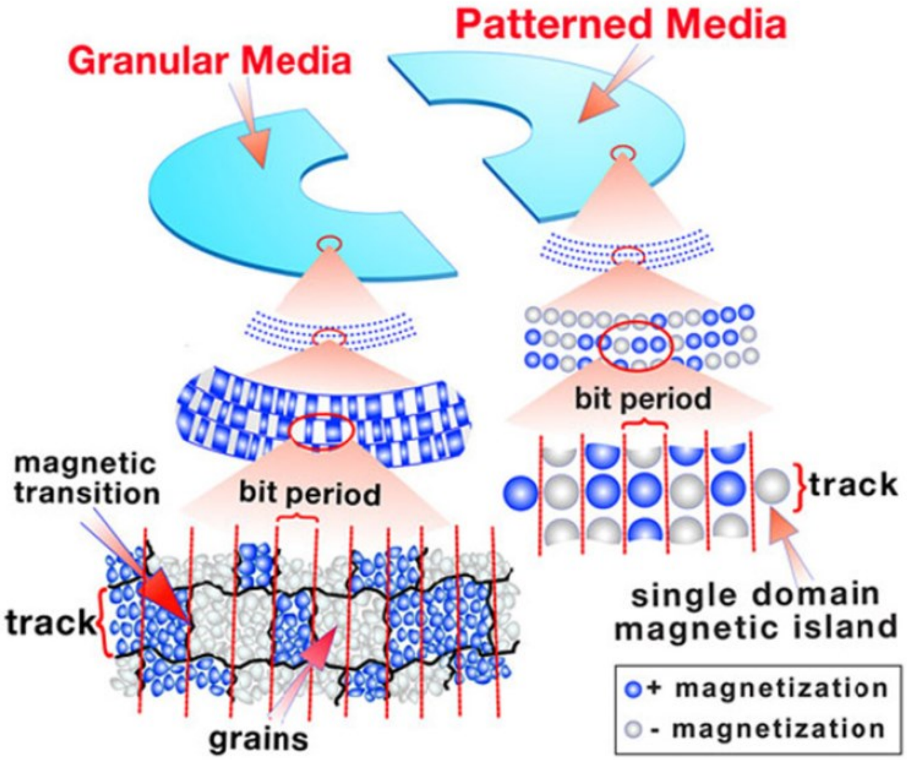
Das wird spannend: Western Digital setzt auf MAMR, Seagate weiter auf HAMR. Ob sich eine der beiden Techniken zur Erhöhung der Festplattenkapazität durchsetzen wird, bleibt sicher noch einige Zeit unklar. Anders als bei den ebenfalls kapazitätssteigernden Techniken Helium-Füllung und Shingled Magnetic Rcording sind aber nun erstmals zwei Unternehmen mit konkurrierender Technik auf dem Vormarsch.

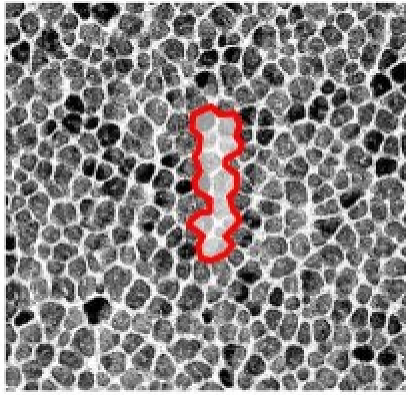
Seagate-Chef Dr. Mark Re wirbt im Seagate-Blog für die HAMR-Technik. Mit HAMR, Heat Assisted Magnetic Recording, seien bereits heute Kapazitäten bis zu 2 TBit pro Quadratzoll möglich. In den letzten neun Jahren hätten die Ingenieure jährlich eine Steigerung von 30 Prozent bei der Datendichte erreicht. 2019 sollen bereits Festplatten mit mindestens 20 TByte möglich sein, für 2023 erwartet Re bereits 40 TByte. Zur Erinnerung: Auch WD verspricht mit seiner Mikrowellen-Technik MAMR-Technik 40-TByte-Festplatten, aber erst zwei Jahre später.

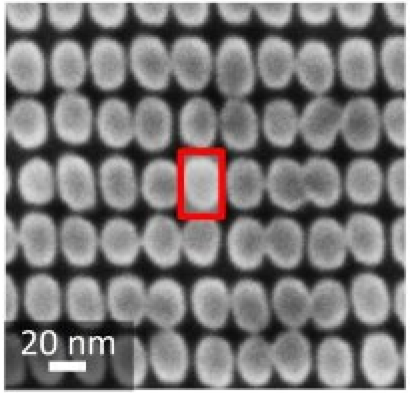
Introduction to Magnetic Materials

Temporal Evolution of Storage Density in HDD









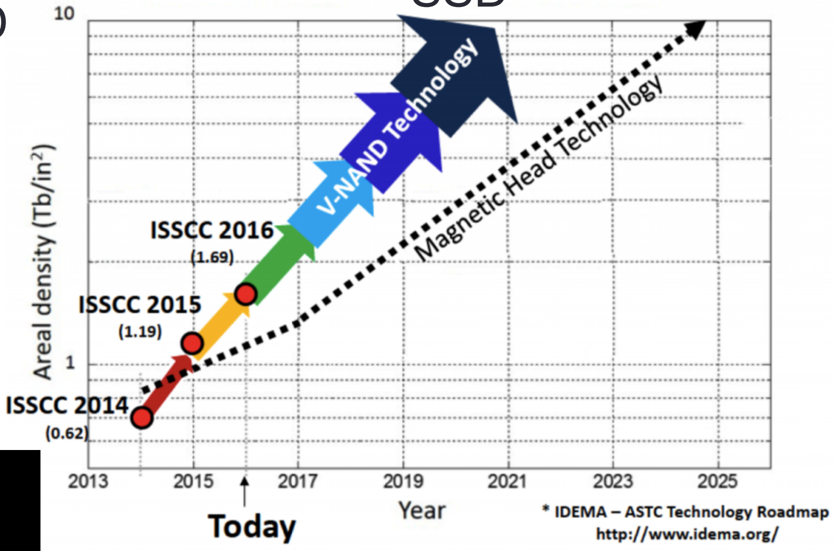
Griffith et al., J. Phys. D: Appl. Phys. 46, 503001 (2013)

Introduction to Magnetic Materials

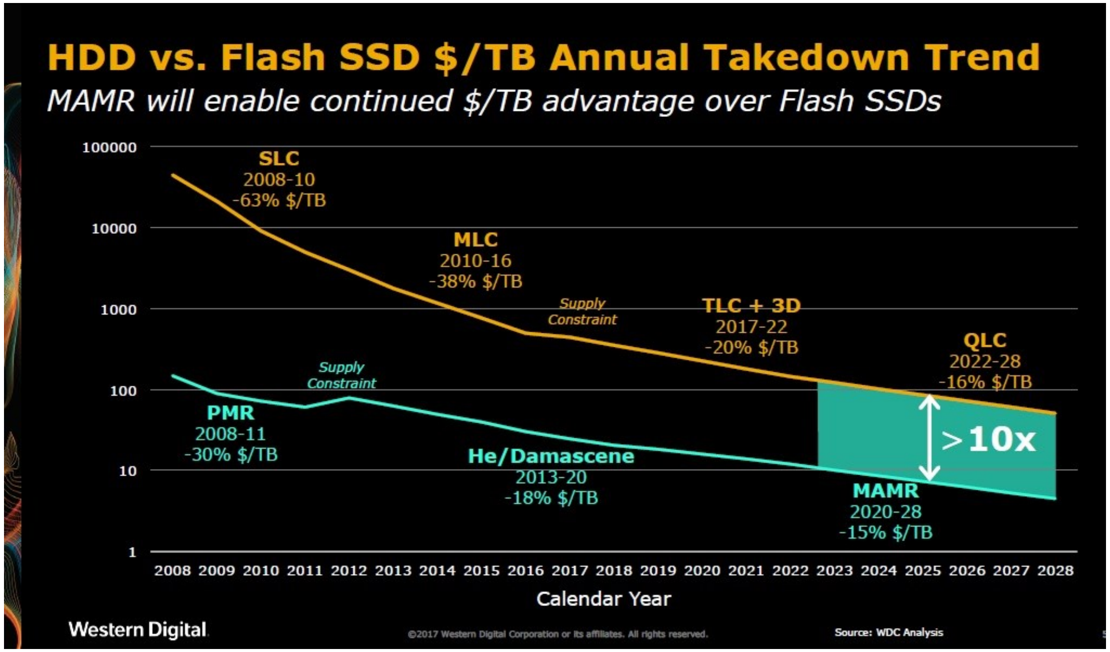
Temporal Evolution of Storage Density in HDD

HDD vs FLASH (SSD) memory

SSD



* IDEMA – ASTC Technology Roadmap
http://www.idema.org/



HDD vs. Flash SSD $/TB Annual Takedown Trend

MAMR will enable continued $/TB advantage over Flash SSDs

Western Digital

Source: WDC Analysis

Is magnetism is out of the game?

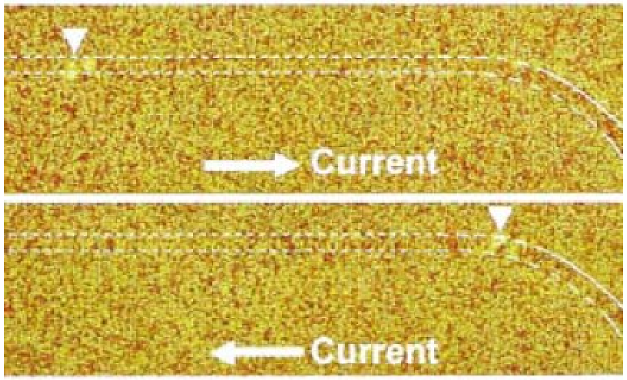
Introduction to Magnetic Materials

New Concepts Triggered by Novel Phenomena

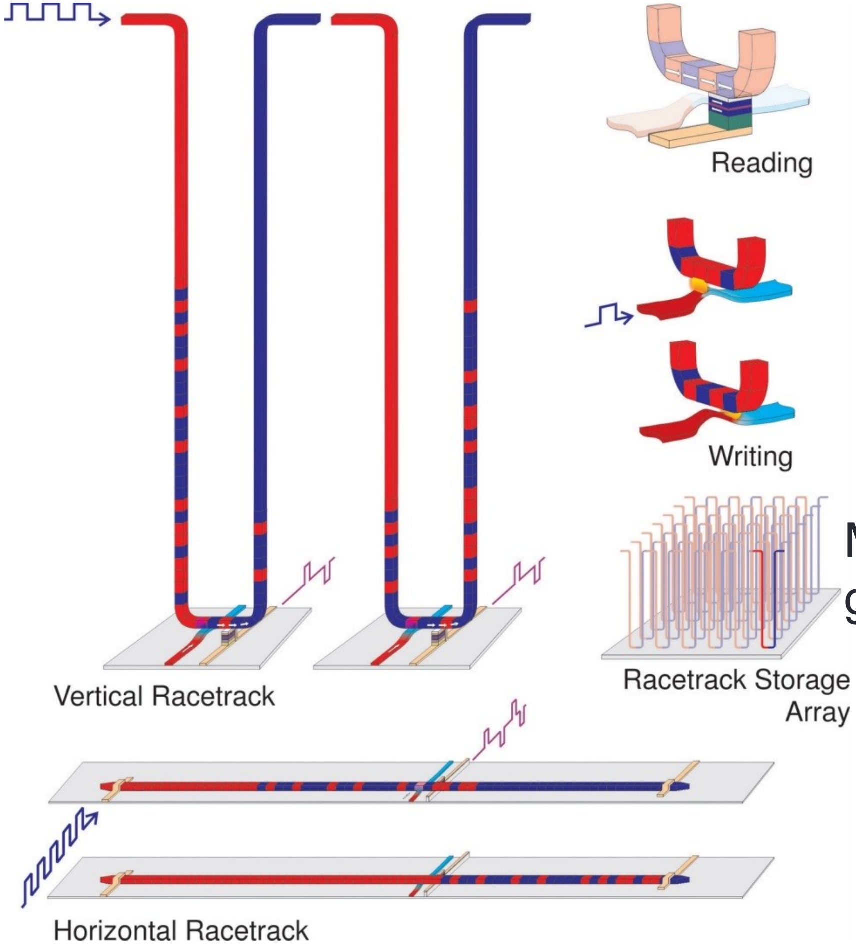
Racetrack Memory (2008)

Electrical currents can manipulate magnetism!

- Oersted fields (1820)
- Current Driven Domain Wall Motion due to Spin-Torque Phenomena (2004)



A. Yamaguchi et al., Phys. Rev. Lett. 92, 077205 (2004)



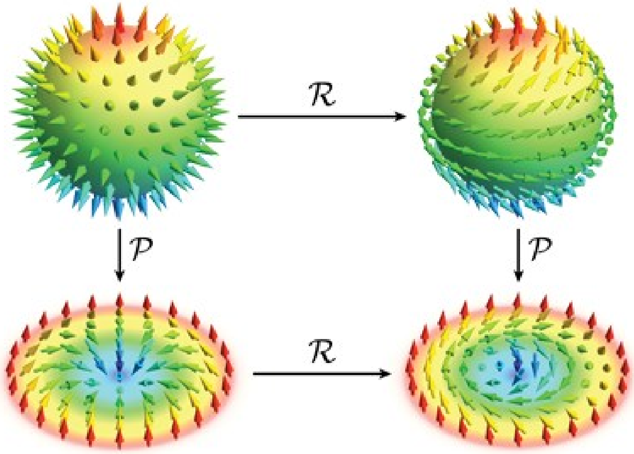
Magnetic storage goes 3D

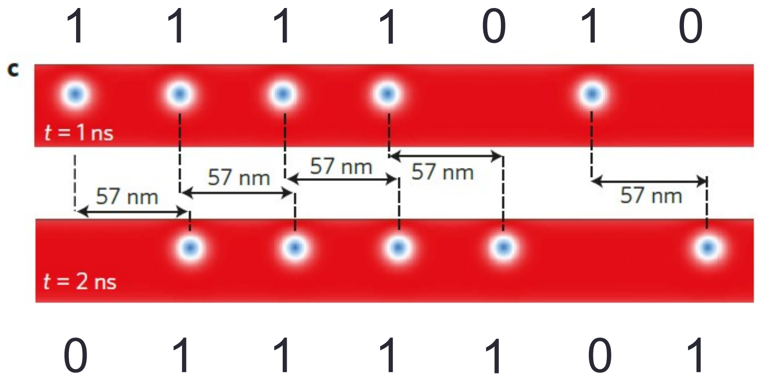
S.S.P. Parkin et al., Science 320, 5873 (2008)

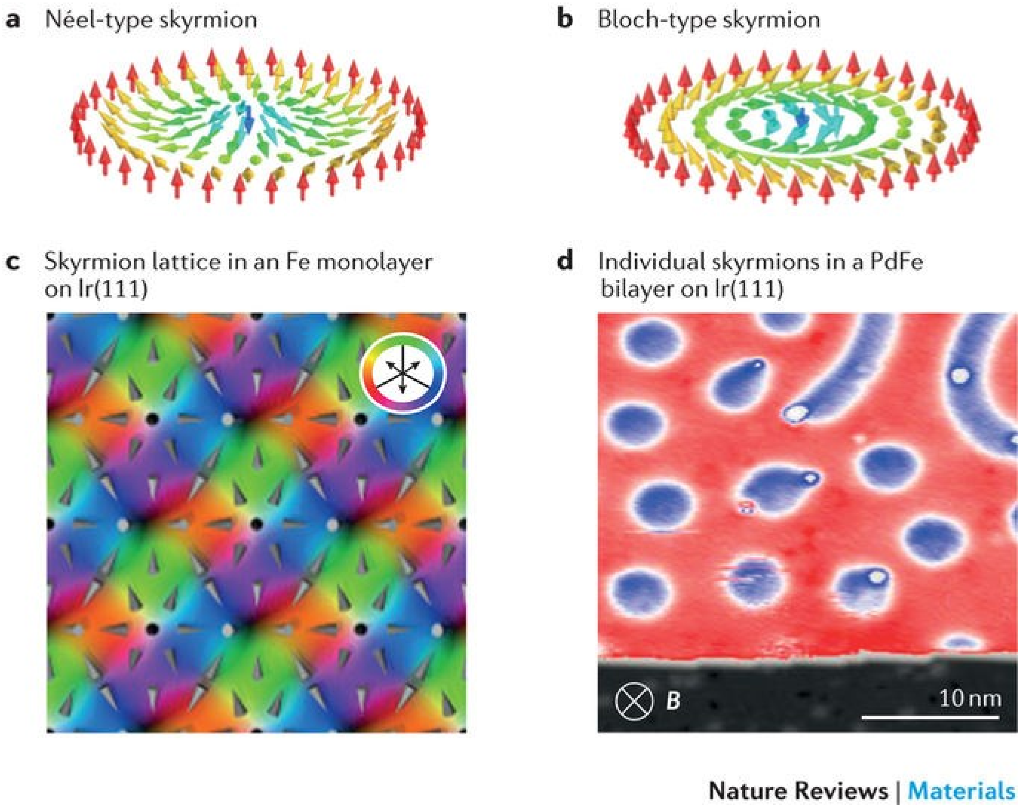
Introduction to Magnetic Materials

New Concepts Triggered by Novel Phenomena

Skyrmion Racetrack Memory (2013)





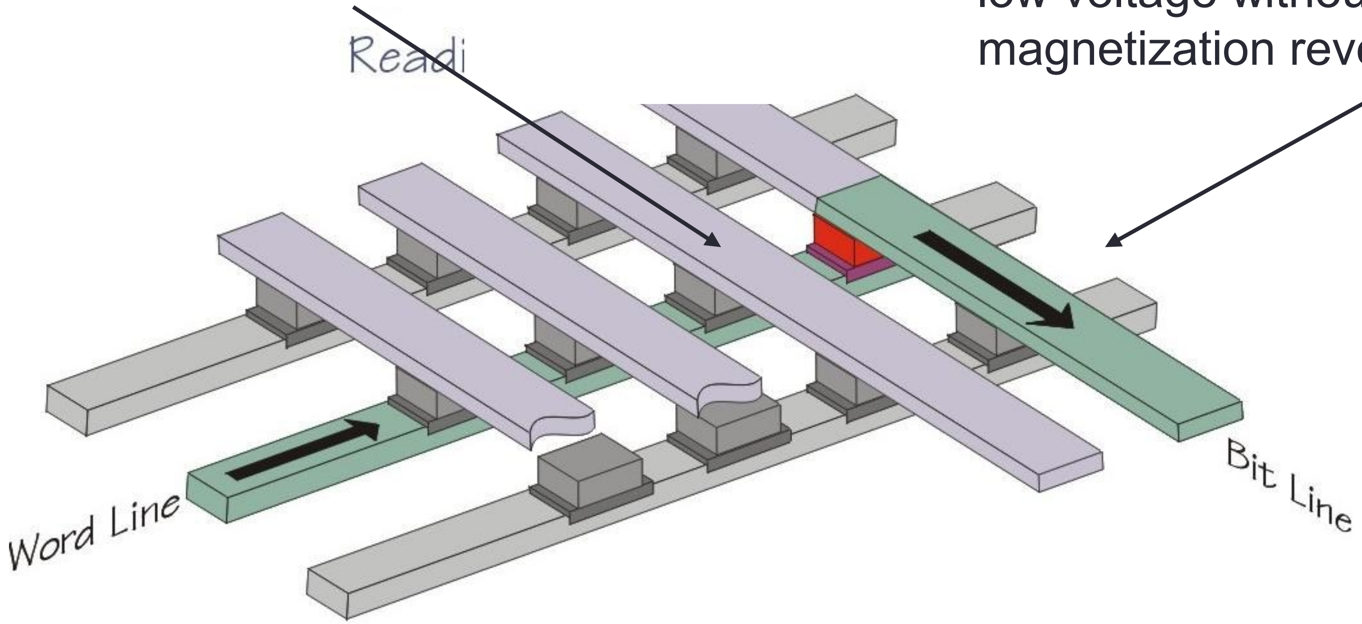


A. Fert et al., Nat. Nanotech. 8, 152 (2013)
A. Fert et al., Nature Rev. Mat. 2, 17031 (2017)

Introduction to Magnetic Materials

New Concepts triggered by novel Phenomena

(Non-Volatile) Magnetic Random Access Memory (MRAM)



Nanoscale:
1dot = 1bit

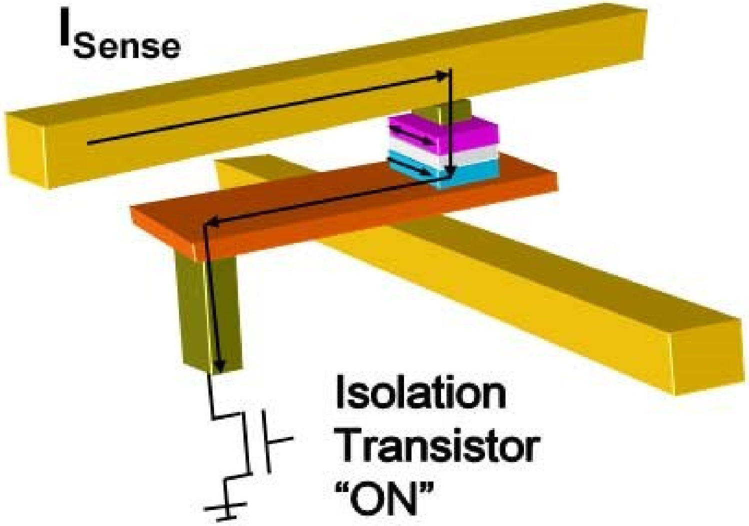
Read by means of
low voltage without
magnetization reversal

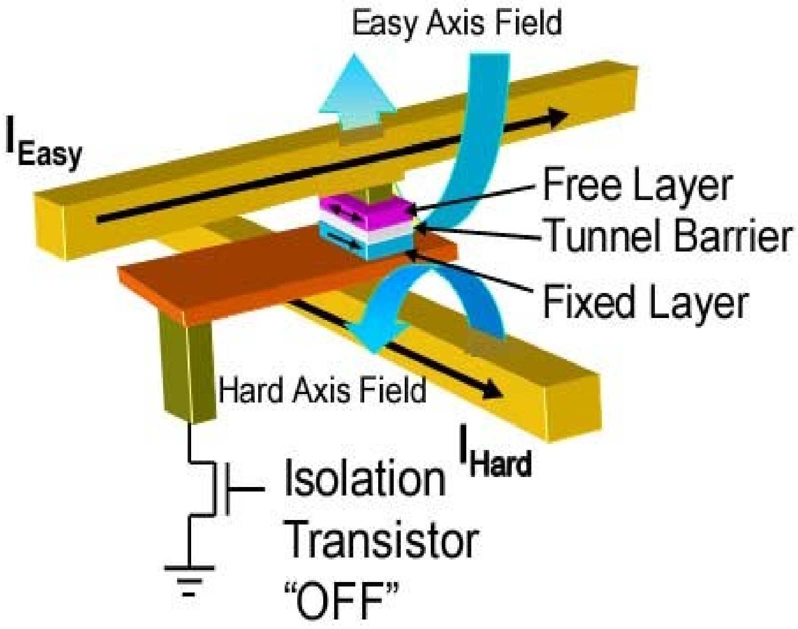
Dot size 20x20nm², distance 20nm: 4 Tbit/inch²

Introduction to Magnetic Materials

New Concepts triggered by novel Phenomena

(Non-Volatile) Magnetic Random Access Memory (MRAM)





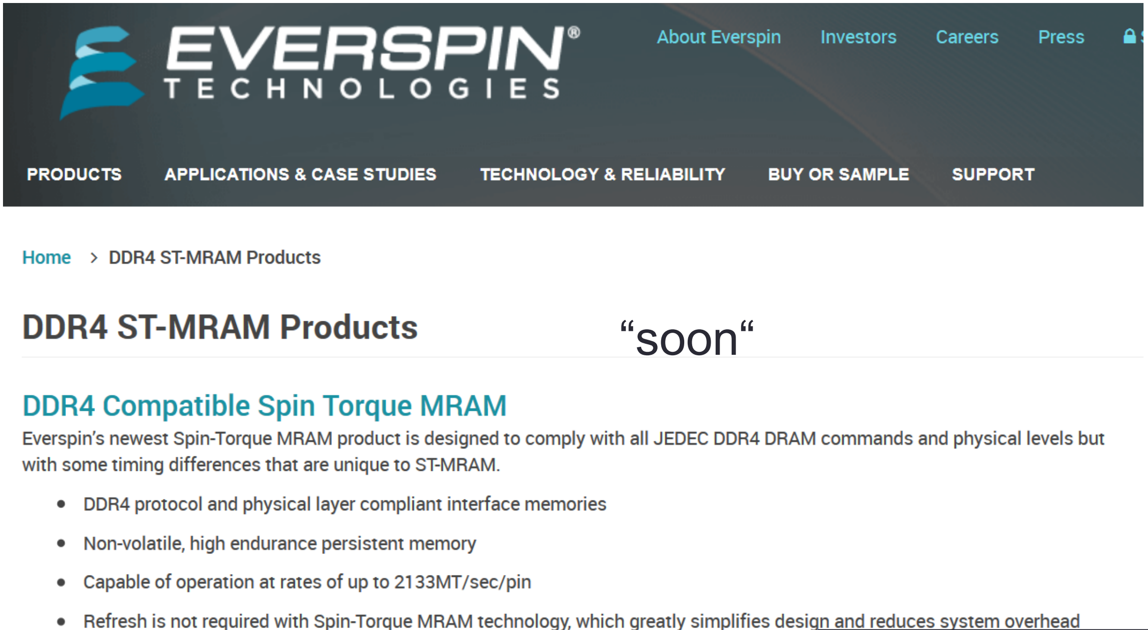
"Read" mode

"Write" mode

Introduction to Magnetic Materials

New Concepts triggered by novel Phenomena

(Non-Volatile) Magnetic Random Access Memory (MRAM)



Home > DDR4 ST-MRAM Products

DDR4 ST-MRAM Products

“soon“

DDR4 Compatible Spin Torque MRAM

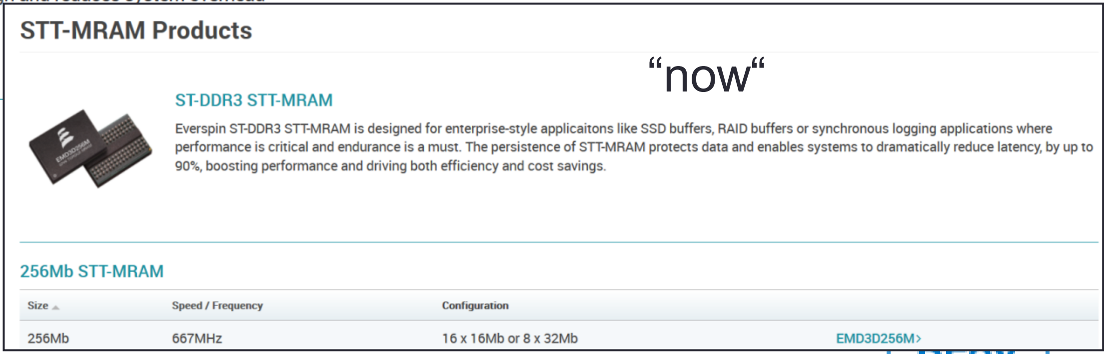
Everspin’s newest Spin-Torque MRAM product is designed to comply with all JEDEC DDR4 DRAM commands and physical levels but with some timing differences that are unique to ST-MRAM.

- DDR4 protocol and physical layer compliant interface memories
- Non-volatile, high endurance persistent memory
- Capable of operation at rates of up to 2133MT/sec/pin
- Refresh is not required with Spin-Torque MRAM technology, which greatly simplifies design and reduces system overhead
- Some unique timing and page size limits

Everspin 1Gb DDR4 Spin-Torque MRAM

The EMD4E001G is a 1 Gigabit ST-MRAM organized as a 128Mb x 8 or 64Mb x 16 and capable of DDR4 operation at rates of up to 2133MT/sec/pin. Product availability will be announced soon.

Request more information about Everspin’s 1Gb ST-MRAM >



STT-MRAM Products

“now“

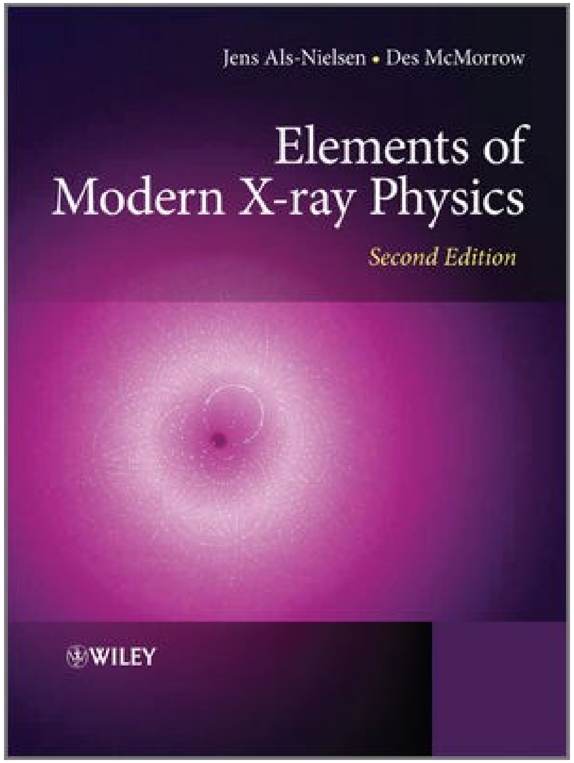
ST-DDR3 STT-MRAM

Everspin ST-DDR3 STT-MRAM is designed for enterprise-style applicaitons like SSD buffers, RAID buffers or synchronous logging applications where performance is critical and endurance is a must. The persistence of STT-MRAM protects data and enables systems to dramatically reduce latency, by up to 90%, boosting performance and driving both efficiency and cost savings.

256Mb STT-MRAM

Size ▲	Speed / Frequency	Configuration	
256Mb	667MHz	16 x 16Mb or 8 x 32Mb	EMD3D256M>

Literature:

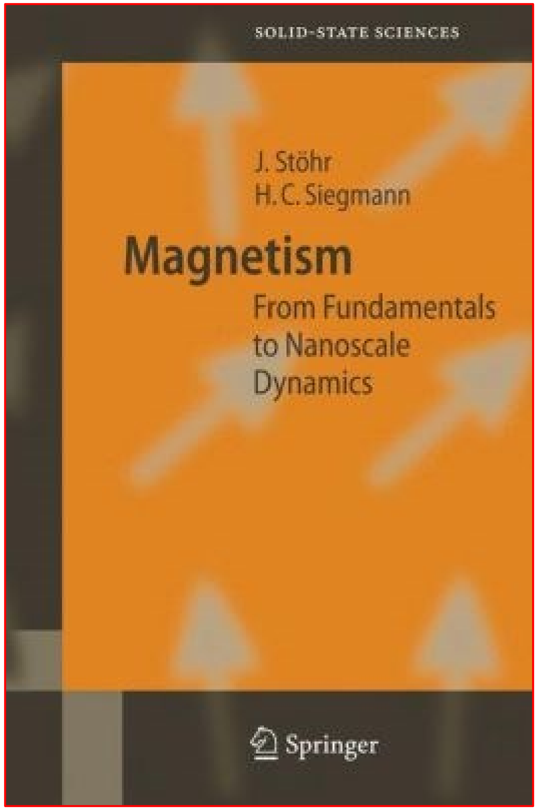


Jens Als-Nielsen • Des McMorrow

Elements of Modern X-ray Physics

Second Edition

WILEY



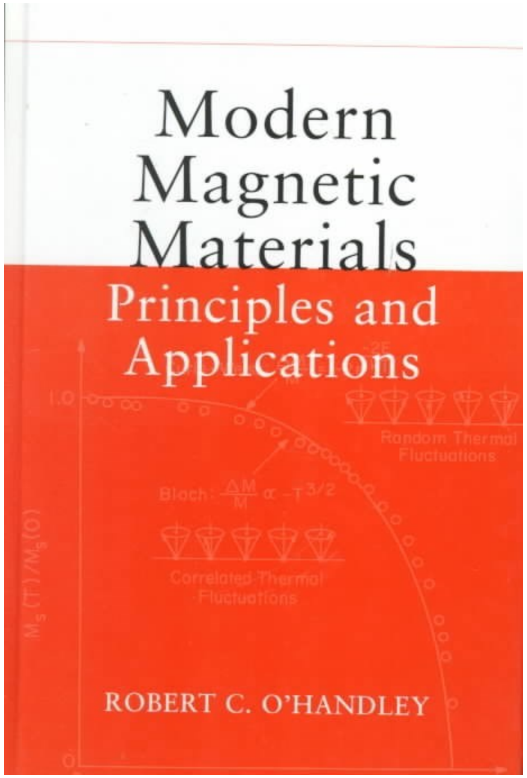
SOLID-STATE SCIENCES

J. Stöhr
H. C. Siegmann

Magnetism

From Fundamentals to Nanoscale Dynamics

Springer



Modern Magnetic Materials

Principles and Applications

ROBERT C. O'HANDLEY

For details about magnetism

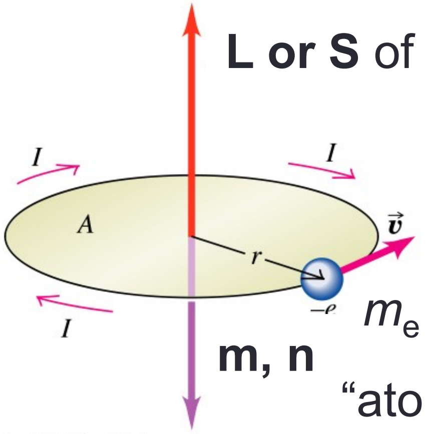
Introduction to Magnetic Materials

1.) Ferromagnetism in a nutshell

- Forms of Magnetic Phenomena
- Contributions to Magnetic Free Energy
- Focus on Systems with Perpendicular Magnetic Anisotropy (Co/Pt multilayers)
- Magnetic Domains and Domain Walls

Ferromagnetism in a nutshell – Forms of magnetism

- Magnetic (dipole) moment **m** (basis element of magnetism)



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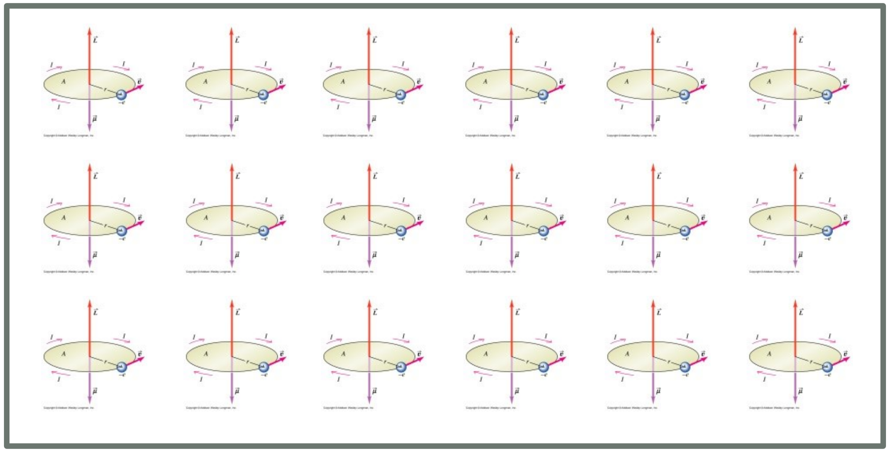
Definition: $\mathbf{m} = I \cdot A \cdot \mathbf{n}$
Unit: $[m] = \mathrm{Am}^2$

“atom“ = conductor (or current) loop (Physik II)

- Magnetization: $\mathbf{M} = \sum \mathbf{m}/V$



Saturation magnetization (“length” of **M**):

$$M_S = |\sum \mathbf{m}|/V$$

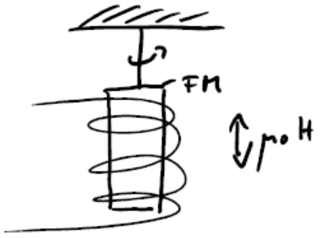
Volume V

Ferromagnetism in a nutshell – Forms of magnetism

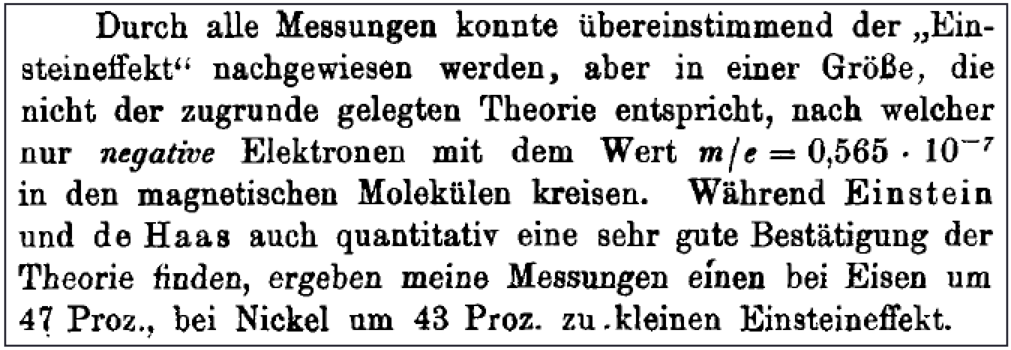
- Connection to angular momentum **L**

  Current loop of moving charges with mass m_e exhibits angular momentum

  $\mathbf{m} = \gamma \mathbf{L}$ γ: gyromagnetic ratio (proportionality proofed 1915 by Einstein-de Haas)



Torsion of string when **M** is changed by magnetic field



Durch alle Messungen konnte übereinstimmend der „Einsteineffekt“ nachgewiesen werden, aber in einer Größe, die nicht der zugrunde gelegten Theorie entspricht, nach welcher nur *negative* Elektronen mit dem Wert $m/e = 0{,}565 \cdot 10^{-7}$ in den magnetischen Molekülen kreisen. Während Einstein und de Haas auch quantitativ eine sehr gute Bestätigung der Theorie finden, ergeben meine Messungen einen bei Eisen um 47 Proz., bei Nickel um 43 Proz. zu kleinen Einsteineffekt.

E. Beck, Ann. Phys. **18**, 1919 (1915)

Gyromagnetic ratio γ :

$$|\vec{m}| = I \cdot A \qquad |\vec{L}| = N m r v$$

N: number of particles

$$\gamma = \frac{IA}{N r m v} = \frac{q \cancel{N} \cancel{\pi} \cancel{r^2} \cancel{T}}{\cancel{N} \cancel{T} r m 2 \cancel{\pi} \cancel{r}}$$

$$v = \frac{2\pi r}{T}$$

$$\boxed{\gamma = \frac{q}{2m}}$$

$$I = \frac{q \cdot N}{T}$$

Ferromagnetism in a nutshell – Forms of magnetism

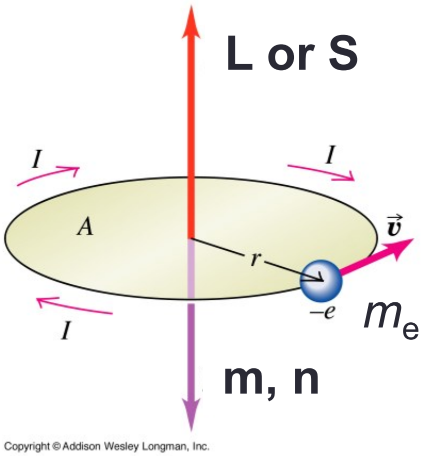
Quantization of angular momentum **L** in units of $\hbar$

➔ Quantization of **m** in units of Bohr magneton μ_B

$$|\vec{m}| = \gamma \hbar = \frac{q\hbar}{2m}$$, for $q = |e|$:

$\mu_B = 9.274 \cdot 10^{-24}\, Am^2$: Bohr Magneton

Note : $\gamma < 0$ for e^- $\quad \vec{L} \uparrow\downarrow \vec{m}$



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Landé- or g- (or gyromagnetic-)factor: $\gamma = g\frac{q}{2m}$

$g = 1$: classical description, angular momentum

$g \simeq 2$ for spin

Ferromagnetism in a nutshell – Forms of magnetism

> Forms of magnetic phenomena in solid states

Diamagnetism and Paramagnetism

- Lorentz-force on moving charges in a magnetic field **B**: $\mathbf{F} = q\mathbf{v} \times \mathbf{B}$

- Two further terms in Hamiltonian: $H = H_0 + H'$

- For one electron on circular loop (“atom”):

$$H' = \overbrace{\frac{q}{2m_\mathrm{e}}}^{-\mathbf{m}}\mathbf{L} \cdot \mathbf{B} + \frac{q^2}{8m_\mathrm{e}}(\mathbf{B} \times \underset{\uparrow}{\mathbf{r}})^2$$

electron radius

1.) Paramagnetic term
- energy of magnetic dipole in field
- alignment of **m** with magnetic field **B**
- *T* dependent (later)

2.) Diamagnetic term
- all materials are diamagnetic
- always > 0
- inhomogeneous field: atom can reduce energy when moving to region of lowest field
- *T* independent

Ferromagnetism in a nutshell – Forms of magnetism

- Forms of magnetic phenomena in solid states

Diamagnetism and Paramagnetism

- Ratio of both corrections:

$$\frac{\frac{q}{2m}\vec{L}\cdot\vec{B}}{\frac{q^2}{8m}|(\vec{B}\times\vec{r})^2|} \geq \frac{\overbrace{\frac{q}{2m}\hbar}^{\mu_B} B}{\frac{q^2}{8m}B^2 r^2} = \frac{4\hbar}{qBr^2} = 10^4 \quad \text{for } B = 10\,T, \; r = 1.5\,\text{Å}$$

- Comparison of paramagnetic term to thermal energy at room temp. for B = 10 T:

$$E_{therm} = 25\text{ meV} \quad (k_B T)$$

$$E_{para} = 0.58\text{ meV}$$

$$E_{para} \ll E_{therm}$$

- Thermodynamic description:

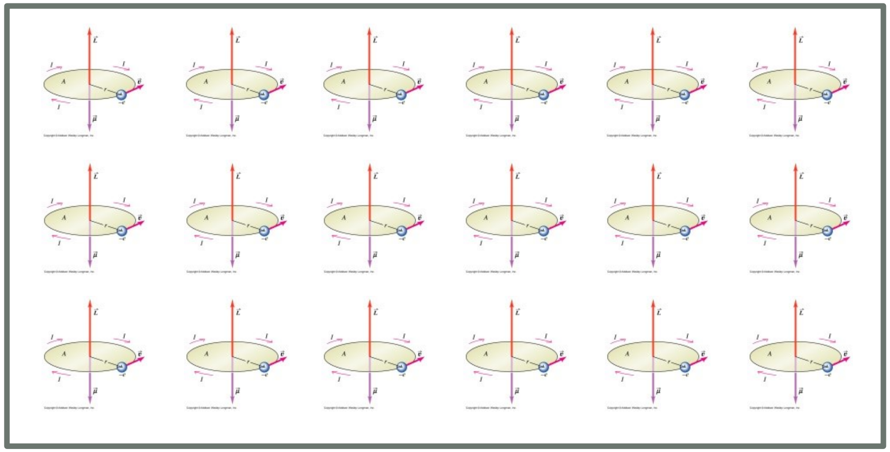
$$\frac{<m>}{\mu_B} = \tanh\frac{\mu_B B}{k_B T} \approx \frac{\mu_B B}{k_B T} = \frac{E_{para}}{E_{therm}} = 0.023 \text{ at } 10\,T$$

Ferromagnetism in a nutshell – Forms of magnetism

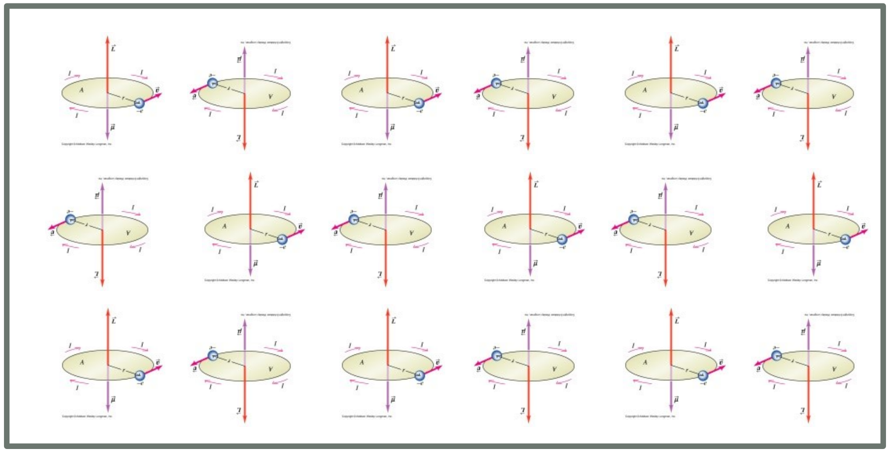
- Forms of magnetic phenomena in solid states

Materials with long range magnetic order (without external magnetic field) due to strong interaction between electron‘s magnetic moments

Ferromagnetism (FM)



Antiferromagnetism (AFM)



Classic description via mean field (Weiß 1907): $|\mathbf{B}_{\mathrm{xc}}| = \mu_0\lambda(J)|\mathbf{M}| = 10^3$ T!

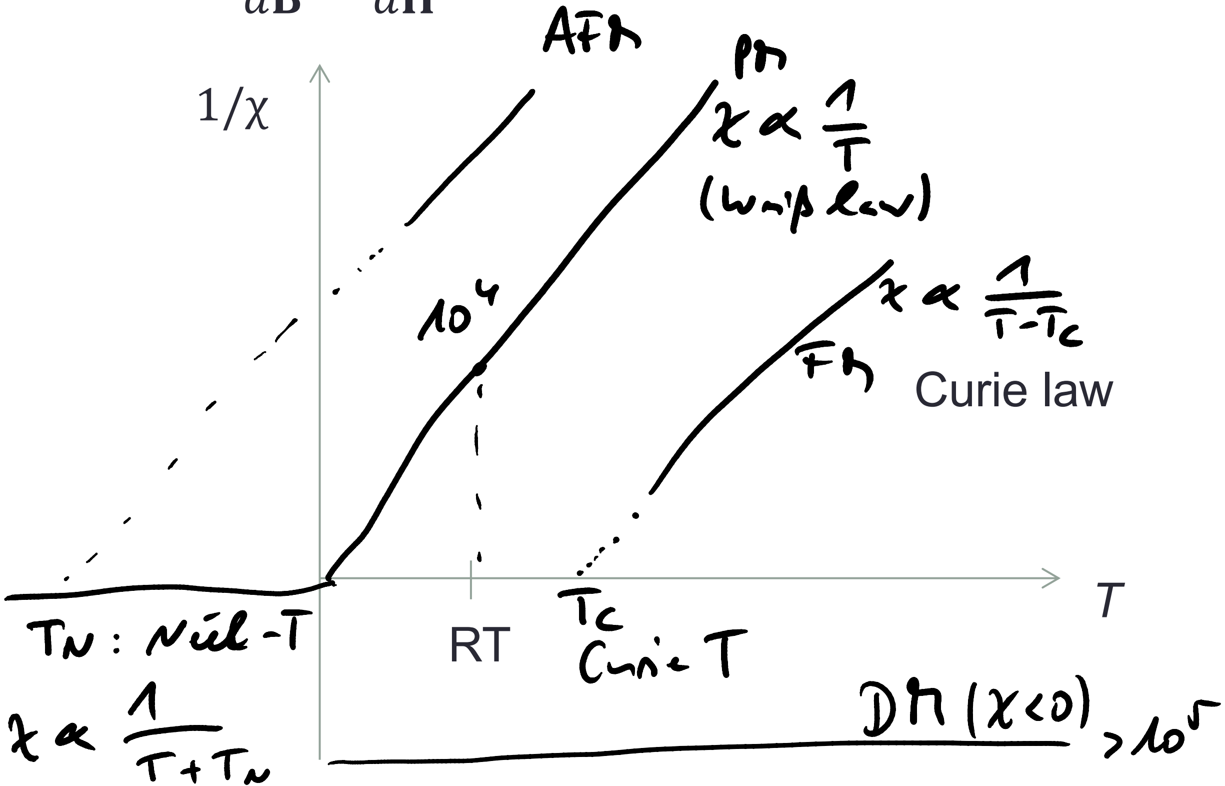
Ferromagnetism in a nutshell – Forms of magnetism

- Forms of magnetic phenomena $\left(M = -\frac{\partial E}{\partial B} \leftarrow \text{free energy}, \ \chi = -\mu_0 \frac{\partial^2 E}{\partial B^2} \right)$

  Classification by means of magnetic susceptibility χ, i.e., response to of magnetization to magnetic field (in high *T* regime):

$$\chi = \mu_0 \frac{d\mathbf{M}}{d\mathbf{B}} = \frac{d\mathbf{M}}{d\mathbf{H}}$$

Magnetic flux density **B**, Magnetic field **H**



<u>Note</u>:
For FM and AFM state (low T)

$\vec{M} \neq \chi \vec{H}$, χ scalar

$\chi = f(\vec{H})$

$\chi = f(\langle hkl \rangle)$

$\chi = f(\text{shape})$

$\chi = f(\text{past})$

Ferromagnetism in a nutshell – Magnetic energies

- Magnetic free energy

$$E_{\text{total}} = E_{\text{XC}} + E_{\text{MCA}} + E_{\text{demag}} + E_{\text{Zeeman}} + E_{\text{DMI}} + \ldots$$

Exchange energy

Magnetocrystalline Anisotropy

Demagnetization or stray field energy

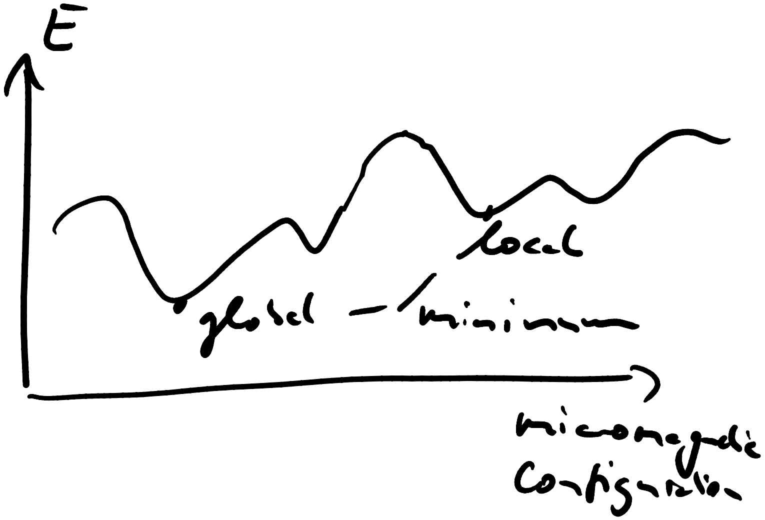
Zeeman energy

Dzyaloshinskii-Moriya interaction

In equlibrium:

$dE/dm_i = 0$

$(d^2E/dm_i^2 > 0)$



Ferromagnetism in a nutshell – Magnetic energies

Exchange energy

- Origin:

  1.) Coulomb interaction between electrons $H_{\text{Coulomb}} = \frac{1}{2}\sum_{i\neq j}\frac{e^2}{4\pi\epsilon_0 r_{ij}}$

  2.) Pauli‘s exclusion principle: Total wave function $|\phi>=|\Psi>\cdot|\chi>$ is antisymmetric when interchanging two identical = undistinguishable particles

Spatial wave function

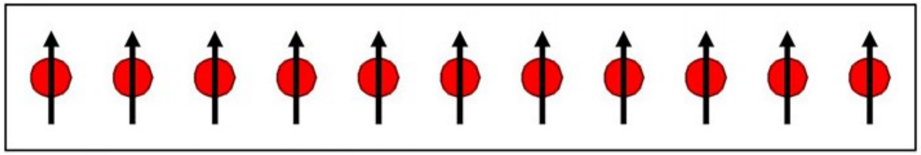
$$J \propto <\Psi_{\text{symmetric}}|H_{\text{Coulomb}}|\Psi_{\text{symmetric}}> - <\Psi_{\text{antisymmetric}}|H_{\text{Coulomb}}|\Psi_{\text{antisymmetric}}>$$

Exchange constant (or integral)

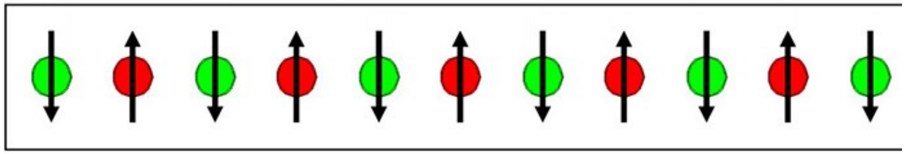
➔ Heisenberg exchange (effective spin-spin interaction)
(generally, only next neighbor interaction)

$$\boxed{E_{\text{XC}} = -\sum_{i\neq j} J_{ij}\,\mathbf{S}_i\cdot\mathbf{S}_j}$$

$J_1 > 0$ ferromagnetic



$J_1 < 0$ antiferromagnetic



Ferromagnetism in a nutshell – Magnetic energies

- Micromagnetic approximation

➔ Define continuous variables like saturation magnetization $M_S = |\sum \mathbf{m}|/V$

➔ Exchange energy $E_{\mathrm{xc}} = A \int_V \left((\nabla m_x)^2 + (\nabla m_y)^2 + (\nabla m_z)^2 \right) dV$ $m_i = \frac{M_i}{M_S}$

A: exchange stiffness
A ≈ 10 pJ/m

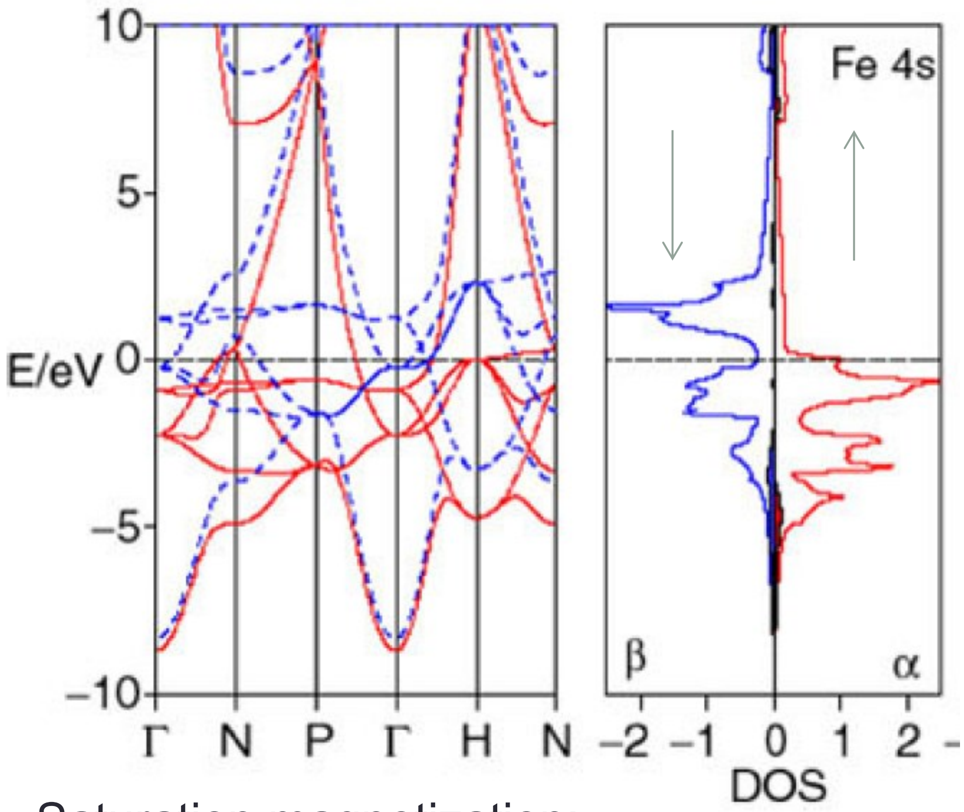
➔ $\Delta E_{xc}/V \approx (E_{\uparrow\downarrow} - E_{\uparrow\uparrow})/V$ = 1 GJ/m³ (0.1 eV/atom ≈ 4 times thermal energy at RT)

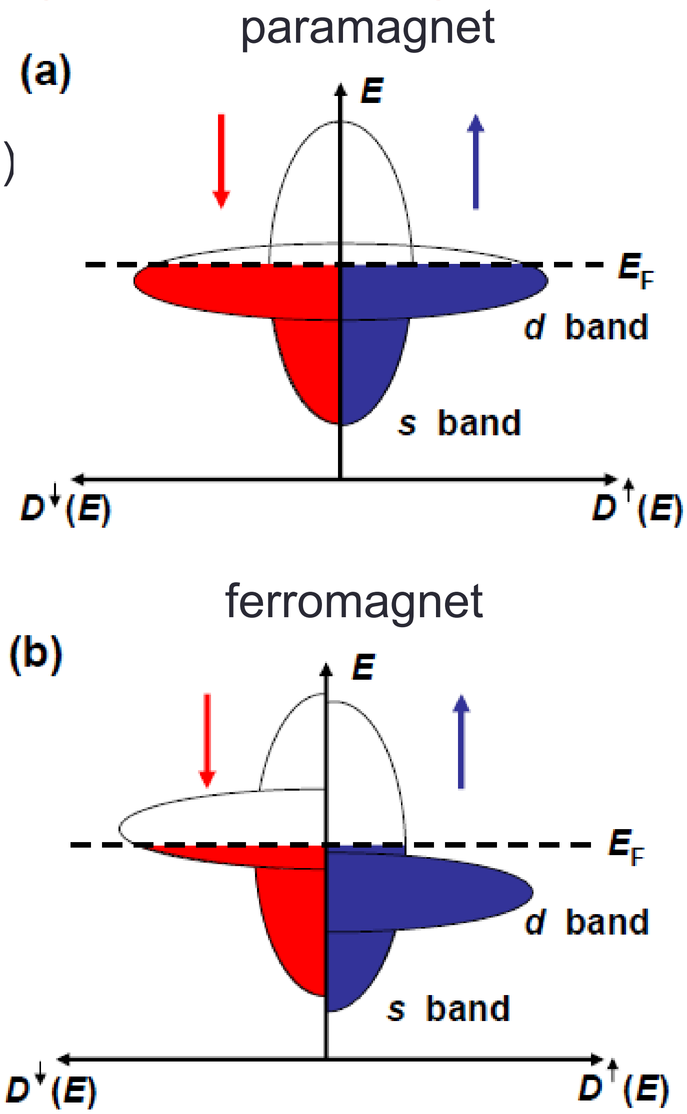
➔ FM at room temperature!

Ferromagnetism in a nutshell – Magnetic energies

- Itinerant (band) Ferromagnetism for Ni, Fe, Co

  (≠ localized FM for rare earth elements like Dy, Tb, Gd)





- Saturation magnetization:

$$M_S = \mu_B(n_\uparrow - n_\downarrow)$$

Ferromagnetism in a nutshell – Magnetic energies

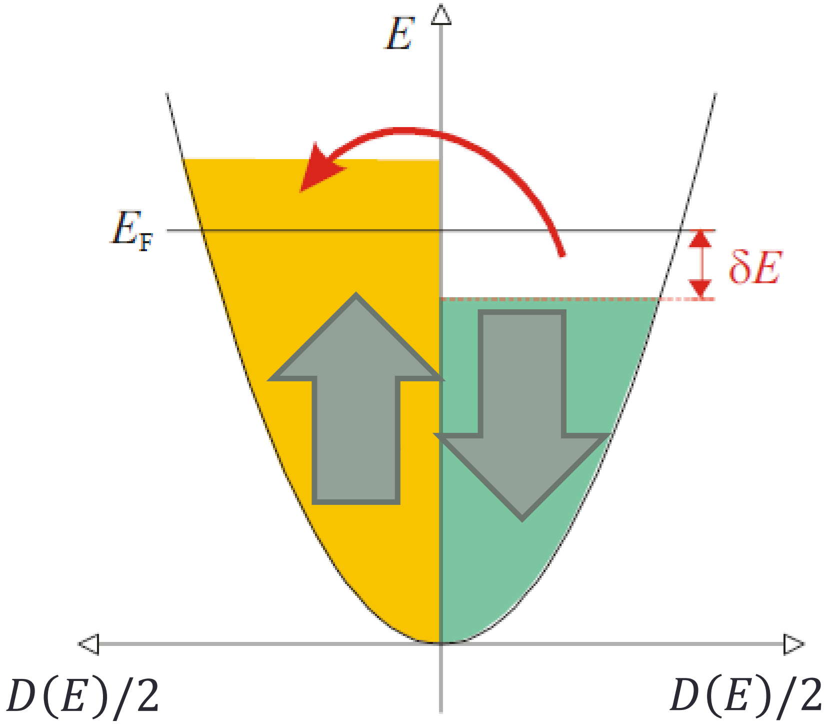
- Itinerant (band) Ferromagnetism for Ni, Fe, Co

  (assume quasi-free electron gas)

  - Saturation magnetization: $M_S = \mu_B(n_\uparrow - n_\downarrow) = \mu_B D(E_F)\delta E$

  Derivation:

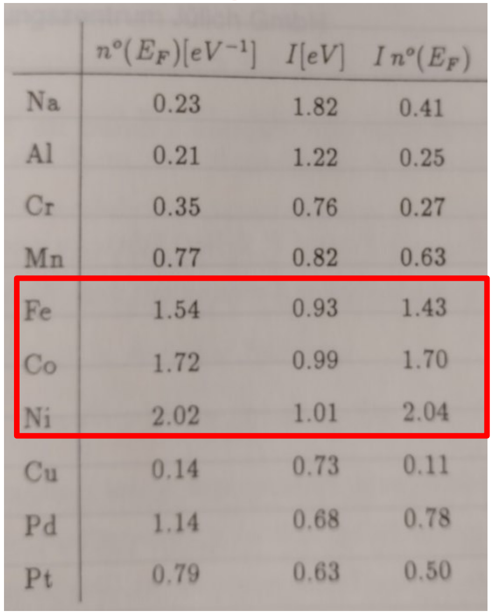
$$n^{\uparrow\downarrow} = \frac{1}{2}\left(n \pm D(E_F)\,\delta E\right)$$



Ferromagnetism in a nutshell – Magnetic energies

- Itinerant (band) Ferromagnetism

$\boxed{\text{Stoner criterion (1939): } I \cdot D(E_\mathrm{F}) > 1}$ I : Stoner parameter



	$n^o(E_F)[eV^{-1}]$	$I[eV]$	$I\,n^o(E_F)$
Na	0.23	1.82	0.41
Al	0.21	1.22	0.25
Cr	0.35	0.76	0.27
Mn	0.77	0.82	0.63
Fe	1.54	0.93	1.43
Co	1.72	0.99	1.70
Ni	2.02	1.01	2.04
Cu	0.14	0.73	0.11
Pd	1.14	0.68	0.78
Pt	0.79	0.63	0.50

- Derivation: Comparison of ferromagnet to paramagnet

1.) Increase of kinetic energy: $\Delta E_\mathrm{kin} = \frac{D(E)\delta E}{2}\,\delta E$

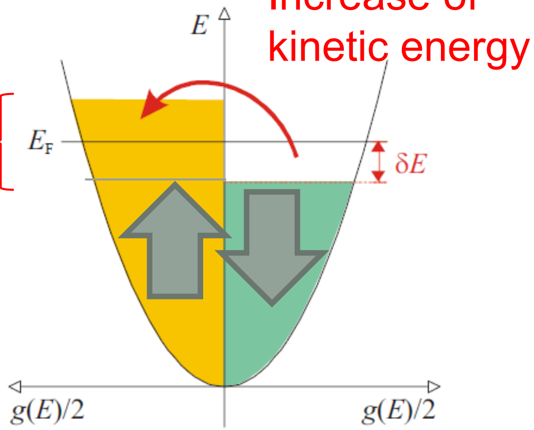
Number of electrons

shift

2.) Decrease of static energy: $dE = -\mu_0 M dH_\mathrm{xc} = -\mu_0 M\lambda(J) dM$

$$\Delta E_\mathrm{pot} = -\int_0^{M_S} M\mu_0\lambda(J)\,\mathrm{d}M = -\frac{\mu_0}{2}\lambda(J){M^2}_S$$

$$= -\frac{1}{2}\underbrace{\lambda(J)\mu_0{\mu_B}^2}_{I}(D(E_\mathrm{F})\delta E)^2$$



Increase of kinetic energy

e- without repulsive Coulomb interaciton

3.) Total energy balance:

$$\Delta E = \Delta E_\mathrm{pot} + \Delta E_\mathrm{kin} = \frac{1}{2}D(E_\mathrm{F})\delta E^2(1 - I \cdot D(E_\mathrm{F}))$$

Ferromagnet if $\Delta E < 0 \Rightarrow \boxed{I \cdot D(E_\mathrm{F}) > 1}$

Ferromagnetism in a nutshell – Magnetic energies

> magnetocrystalline anisotropy

- Gedankenexperiment:
  Assume an infinite (a) amorphous material/ (b) crystal, which orientation has **M**?

(a) All spins are aligned in parallel (**M** exists) due to exchange interaction but the direction of **M** is fluctuating

(b) Crystal field theory: Crystal order breaks isotropy
+ (Quenched) orbital momentum **L** is firmly linked to crystal lattice
+ **Spin orbit interaction** proportional to $\mathbf{L} \cdot \boldsymbol{S}$

➔ Energy depends on orientation of **M** with respect to the crystal axes
= magnetocrystalline anisotropy

Ferromagnetism in a nutshell – Magnetic energies

- magnetocrystalline anisotropy

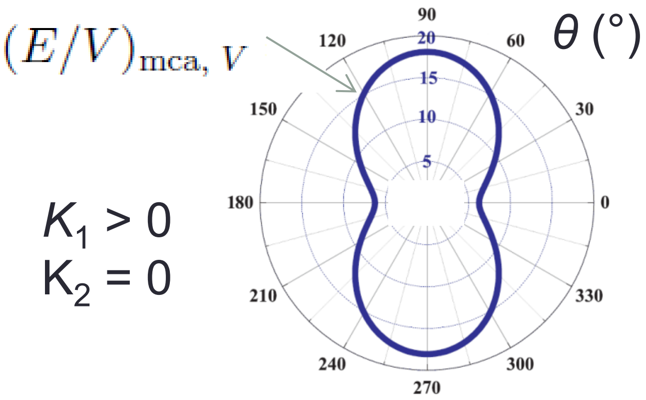
  - Most simple case: *Uniaxial* MCA like in hcp crystals (e.g. Co at room temperature)

$$(E/V)_{\text{mca, }V} = K_{1V}\sin^2\theta + \underline{K_{2V}\sin^4\theta} + \mathcal{O}(\sin^6\theta)$$

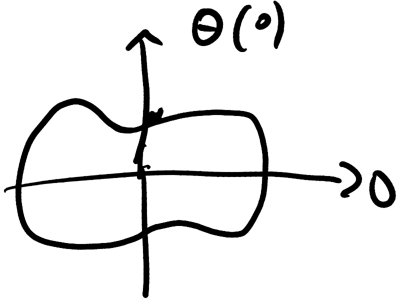
Higher order is also considered in today's excercise (15.5.18)

$K_{1V,\text{ Co}}$ = +0.5 MJ/m³ (three orders of magnitude smaller than XC)

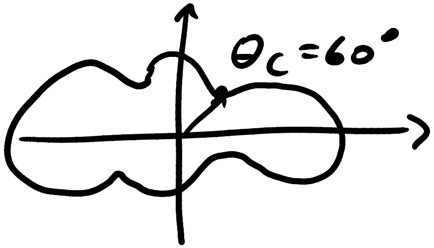
➔ The (0001) axis is the „easy axis of magnetization“ for Co



$K_1 > 0$
$K_2 = 0$



$k_1 < 0$
$k_2 = 0$



$-2k_2 < k_1 < 0$
$k_2 > 0$

See today's exercise

- Note: Magnetoelastic anisotropy due to lattice strain yields to higher anisotropy constants, e.g. KV = 2.5 MJ/m³ for tetragonally distorted FePt $L1_0$ alloys

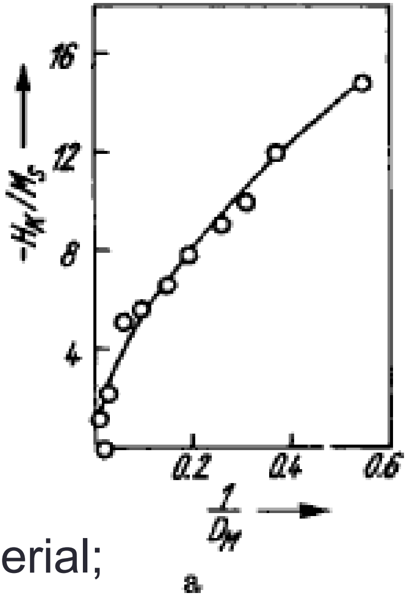
Ferromagnetism in a nutshell – Magnetic energies

- (magnetocrystalline) interface anisotropy (Néel‘s pair interaction model 1959)

  - origin: Symmetry breaking at interface as atoms at interfaces have less nearest neighbors of the same element

$$(E/V)_{\text{mca},\,S} = \frac{2K_S \sin^2\theta}{t}$$

$$\boxed{E_{\text{MCA,total}}/V = (K_V + 2K_S/t)\sin^2\theta}$$



- Discovery by Gradmann and Müller for NiFe(111) on Cu (1968)

- Strongly depends on interface orientation and paramagentic material; high positive value for Co(0001)/Pt(111); discovered 1988:

$K_{S,\,\text{Co/Pt}}$ = +1 mJ/m² ~10 MJ/m³ (two orders of magnitude smaller than XC interaction)

↑ when considering half atomic layer (1 Å)

➔ The (0001) axis is the „easy axis of magnetization“ for Co/Pt for small *t*

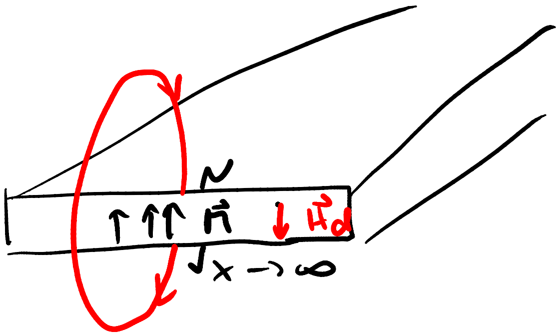
Ferromagnetism in a nutshell – Magnetic energies

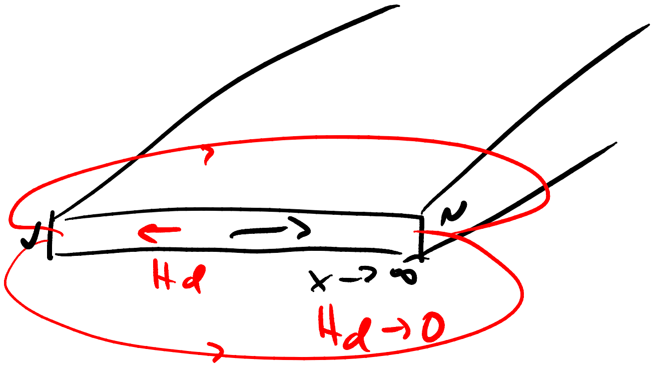
- Demagnetization energy E_d (shape anisotropy)

  - Gedankenexperiment II:
  What happens when cutting out a thin slice of an infinite ferromagnet?

  ➔ (crystalline materials: magnetocrystalline interface anisotropy)

  ➔ Generation of surface charges and demagnetization energy (positive definite) when **M** has components along surface normal
  ➔ **M** prefers to align along the surface (pole avoidance principle)
  ➔ (again) “easy and hard axis of magnetization“





Ferromagnetism in a nutshell – Magnetic energies

$$\vec{B} = \mu_0 (\vec{H} + \vec{M})$$

- Demagnetization energy E_d

=Consequence of Maxwell equation: $\boxed{\mathrm{div}\mathbf{B} = \mu_0 \mathrm{div}(\mathbf{M} + \mathbf{H}_\mathrm{d}) = 0}$

$$E_\mathrm{ms} = -\frac{\mu_0}{2} \int\limits_V \mathbf{M} \cdot \mathbf{H}_\mathrm{d} \; \mathrm{d}V$$

Magnetic volume and surface charges

$$\mathbf{H}_\mathrm{d}(\mathbf{r}) = \frac{1}{4\pi} \int_V \frac{(\mathbf{r}-\mathbf{r}')\mathrm{div}\mathbf{M}(\mathbf{r}')}{|\mathbf{r}-\mathbf{r}'|^3} dV' + \frac{1}{4\pi} \oint_{\delta V} \frac{(\mathbf{r}-\mathbf{r}')\mathbf{M}(\mathbf{r}') \cdot \mathbf{n}}{|\mathbf{r}-\mathbf{r}'|^3} dS'$$

Rotational ellipsoids (single domain state): $\mathbf{H}_\mathrm{d} = -\overleftrightarrow{N} \cdot \mathbf{M}$

Symmetry considerations:

$$\overleftrightarrow{N}_\mathrm{sphere} = \begin{pmatrix} \frac{1}{3} & 0 & 0 \\ 0 & \frac{1}{3} & 0 \\ 0 & 0 & \frac{1}{3} \end{pmatrix}, \quad \overleftrightarrow{N}_\mathrm{wire} = \begin{pmatrix} 0 & 0 & 0 \\ 0 & \frac{1}{2} & 0 \\ 0 & 0 & \frac{1}{2} \end{pmatrix}, \quad \overleftrightarrow{N}_\mathrm{film} = \begin{pmatrix} 0 & 0 & 0 \\ 0 & 0 & 0 \\ 0 & 0 & 1 \end{pmatrix}$$

Isotrop ➔ No shape anisotropy

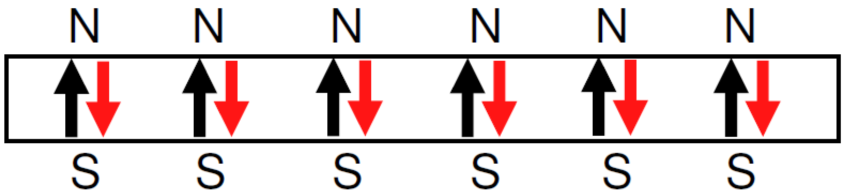
cylindrical wire ~ “cigar“

Ferromagnetism in a nutshell – Magnetic energies

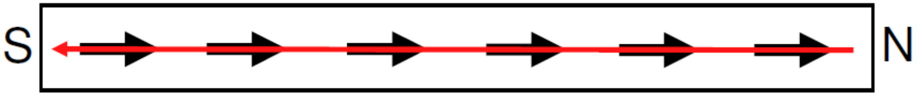
- Demagnetization energy Ed

$$(E/V)_{\text{d, film}} = \frac{\mu_0}{2}(\overleftrightarrow{N}_{\text{film}} \cdot \mathbf{M}) \cdot \mathbf{M} = \frac{\mu_0}{2} M_z^2 = \frac{\mu_0}{2} M_S^2 \cos^2 \Theta$$

$E_{\text{demag}} = \mu_0/2\ M_s^2$



$E_{\text{demag}} \approx 0$



Redefinition of zero: $(E/V)_{\text{d, film}} = \frac{\mu_0}{2} M_S^2 \cos^2 \Theta = -\frac{\mu_0}{2} M_S^2 \sin^2 \Theta + \text{const.}$

$$\boxed{(E/V)_d = -\mu_0/2\ M_s^2 \sin^2 \theta = K_d \sin^2 \theta}$$

For Co at room temperature: M_S =1.44 MA/m →

$K_d = -\mu_0 M_S^2 /2 = -1.3$ MJ/m³

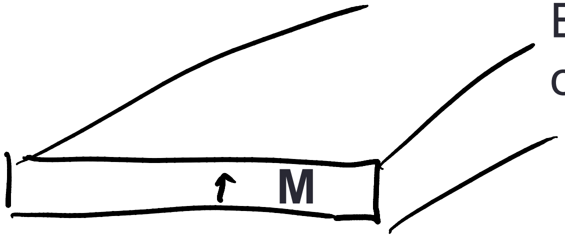
Ferromagnetism in a nutshell – Magnetic energies

- Effective anisotropy constant for uniaxial thin films:

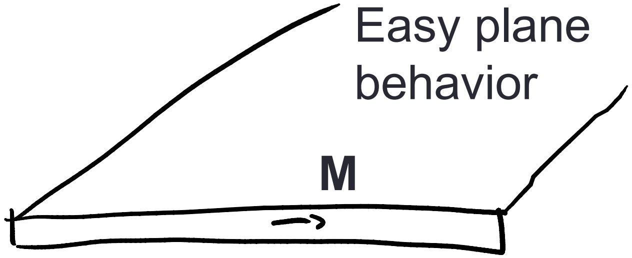
$$K_{1,\mathrm{eff}} = \underbrace{K_{1V} - \frac{\mu_0}{2} M_{\mathrm{S}}^2}_{K_{1V,\mathrm{eff}}} + \frac{2K_{1S}}{t}$$

For Co(0001)/Pt(111) system:

K_{d} = -1.3 MJ/m³
$K_{1\mathrm{V}}$ = +0.5 MJ/m³
$K_{1\mathrm{S}}$ = +1 mJ/m²



Easy axis
out-of-plane!



Easy plane
behavior

➔ $K_{1,\mathrm{eff}} > 0$ for $t < 2$ nm !!!

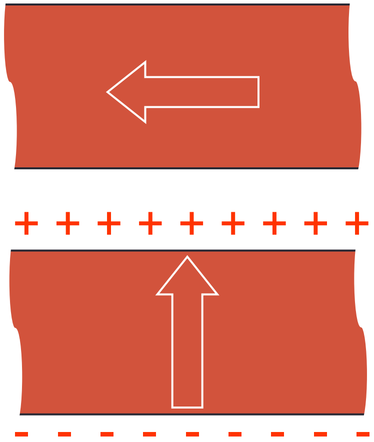
$K_{1,\mathrm{eff}} < 0$ for $t > 2$ nm

Ferromagnetism in a nutshell – Magnetic energies

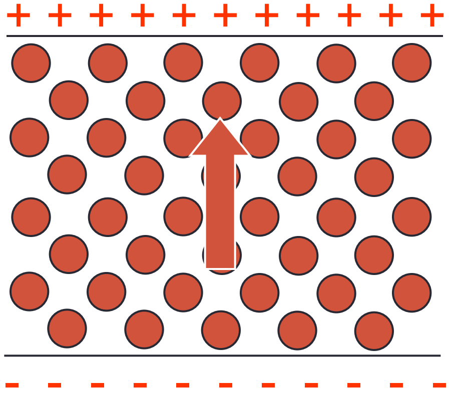
How to realize a perpendicular magnetic anisotropy in a thin film (Summary)?

Lower State "costs" $\mu_0 M_S{}^2/2$

Shape anisotropy



MCA (volume)



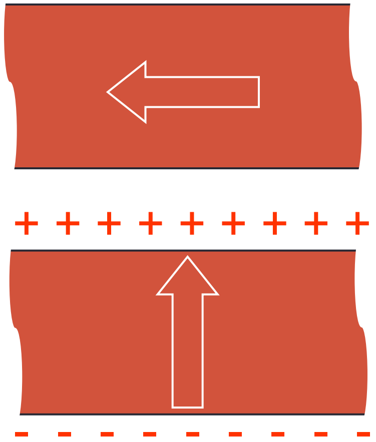
$$K_{eff} = \underbrace{-\frac{\mu_0}{2}M_S^2 \qquad +K_V}_{>0,\text{ e.g. for FePt L}1_0\text{ alloys}}$$

Ferromagnetism in a nutshell – Magnetic energies

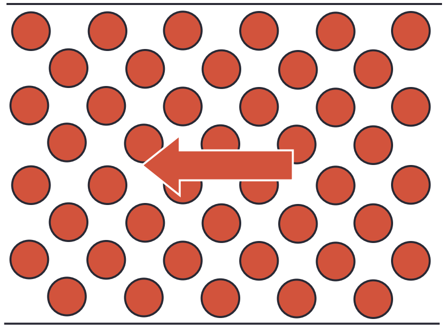
How to realize a perpendicular magnetic anisotropy in a thin film (Summary)?

Lower State "costs" $\mu_0 M_S^{\,2}/2$

Shape anisotropy



MCA (volume)



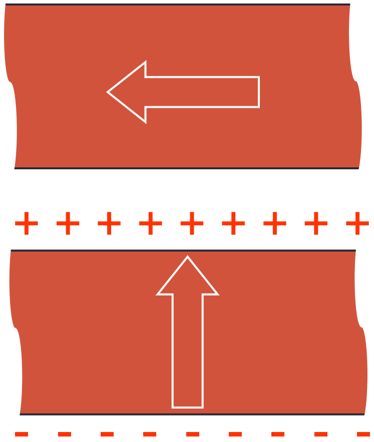
$$K_{eff} = \underbrace{-\frac{\mu_0}{2}M_S^2 \qquad\qquad + K_V}_{<0,\text{ e.g. for Co}}$$

Ferromagnetism in a nutshell – Magnetic energies

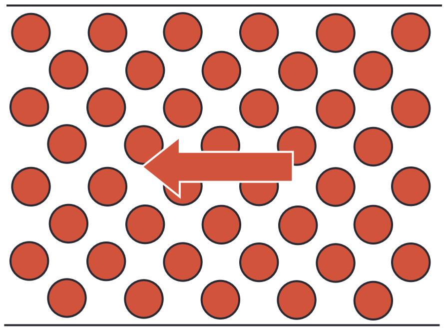
How to realize a perpendicular magnetic anisotropy in a thin film (Summary)?

Lower State “costs” $\mu_0 M_S^2/2$

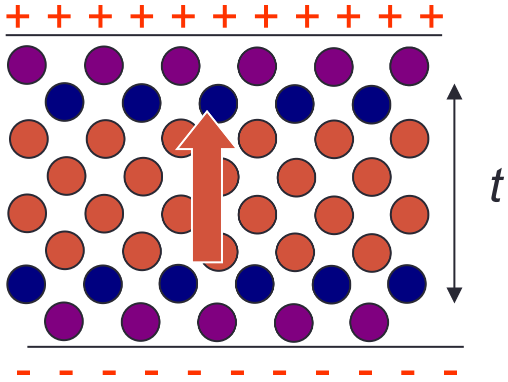
Shape anisotropy



MCA (volume)



MCA (interface)

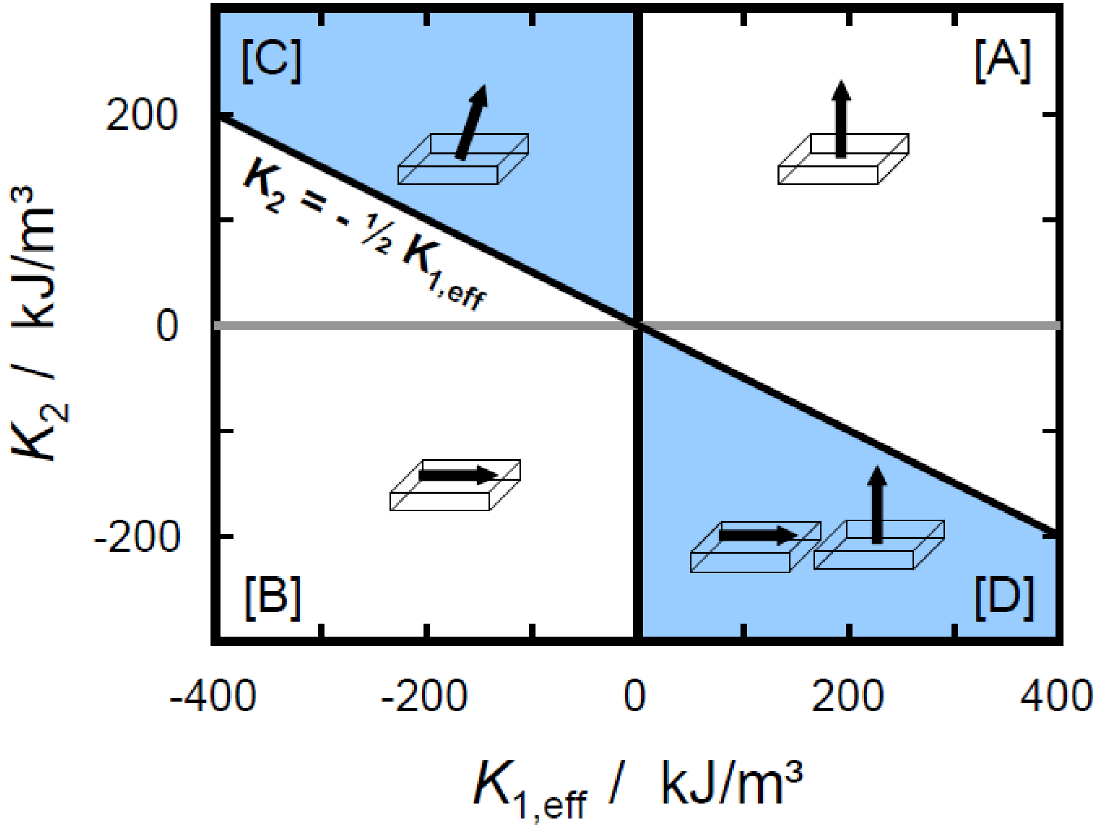


$$K_{eff} = \underbrace{-\frac{\mu_0}{2}M_S^2 + K_V + \frac{2K_S}{t}}$$

>0, e.g. for **Co/Pt,** if $t < 2$ nm

Ferromagnetism in a nutshell – Magnetic energies

- Phase diagram (considering higher orders in anisotropy constants; today‘s excercise)



Ferromagnetism in a nutshell – Magnetic energies

- Zeeman energy

=Energy of magnetization **M** in external magnetic field **H**

$$(E/V)_{\mathrm{Z}} = -\mu_0 \mathbf{M} \cdot \mathbf{H}$$

- Total energy of single domain system (today's Übung):

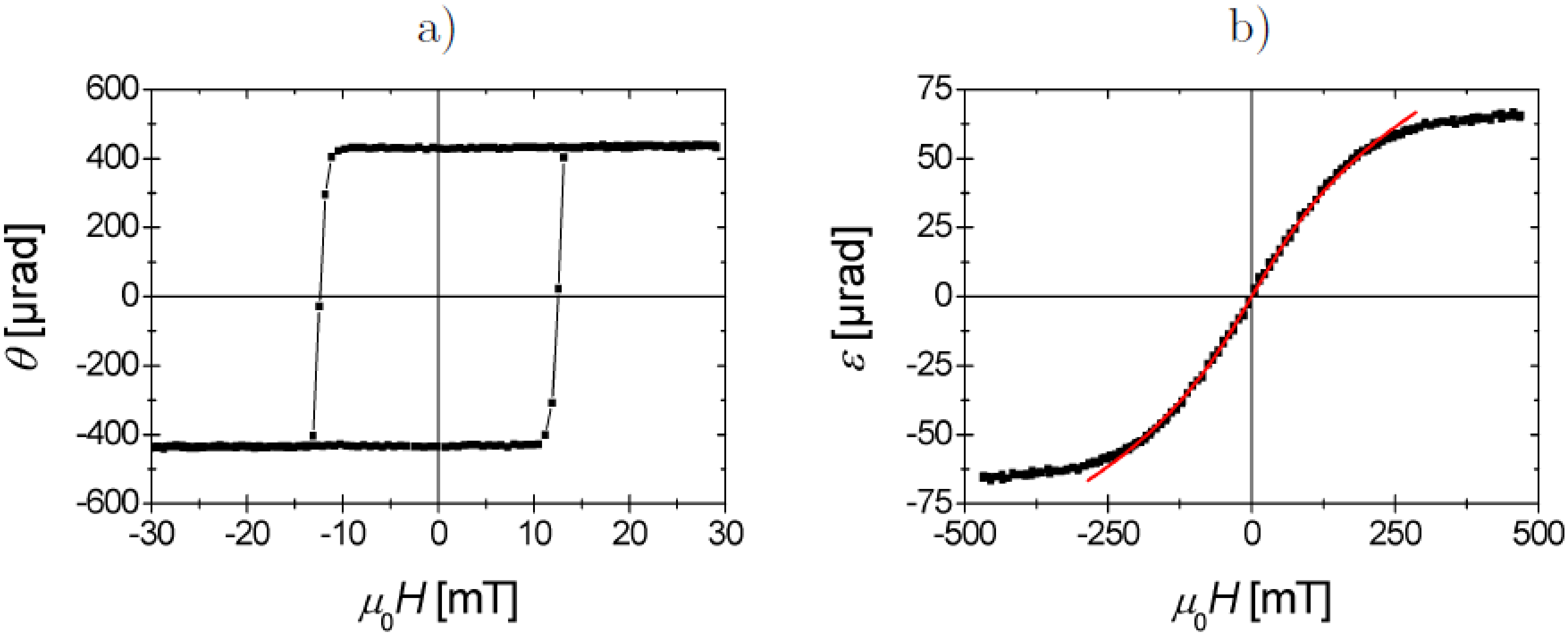
$$E/V = \underbrace{K_{1,\mathrm{eff}} \sin^2\Theta + K_2 \sin^4\Theta}_{\text{MCA+shape anisotropy terms}} - \mu_0 H M_{\mathrm{S}} \cos\Phi$$

Zeeman term

Determination of

Ferromagnetism in a nutshell – Magnetic energies

- Magnetic hysteresis curves

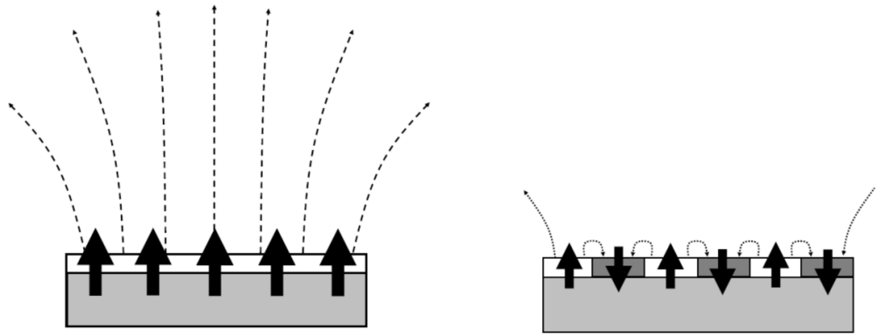


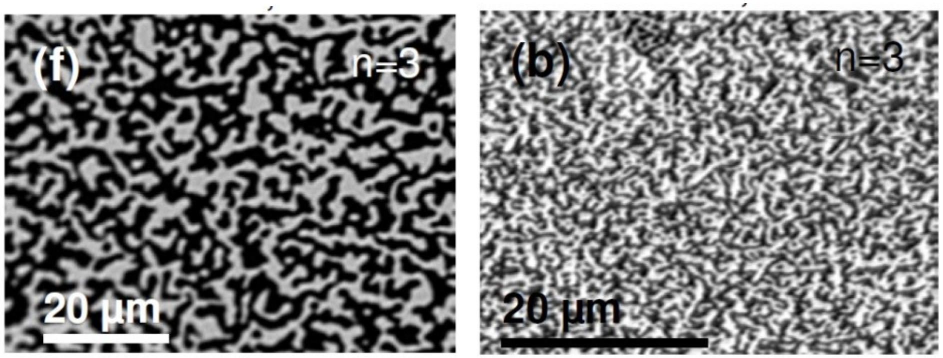
Easy axis
(domain nucleation and domain wall motion)

Hard axis
(coherent rotation of magnetization, today's excercise)

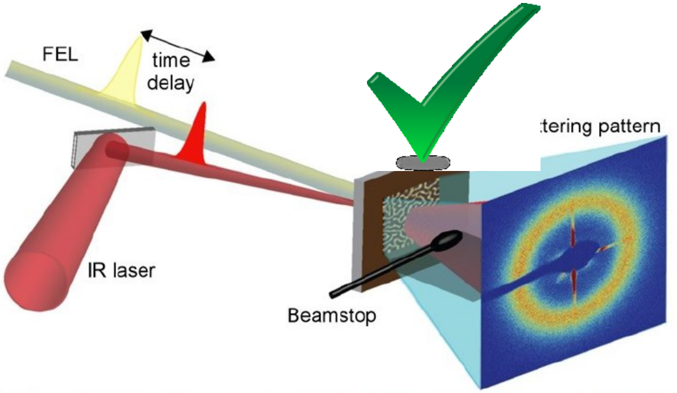
Ferromagnetism in a nutshell – Domains and Walls

- Magnetic domains and domain walls





- Domain walls cost exchange E_{XC} and magnetocrystalline anisotropy energy E_{MCA}
- But: Domain formation reduces stray field energy E_d

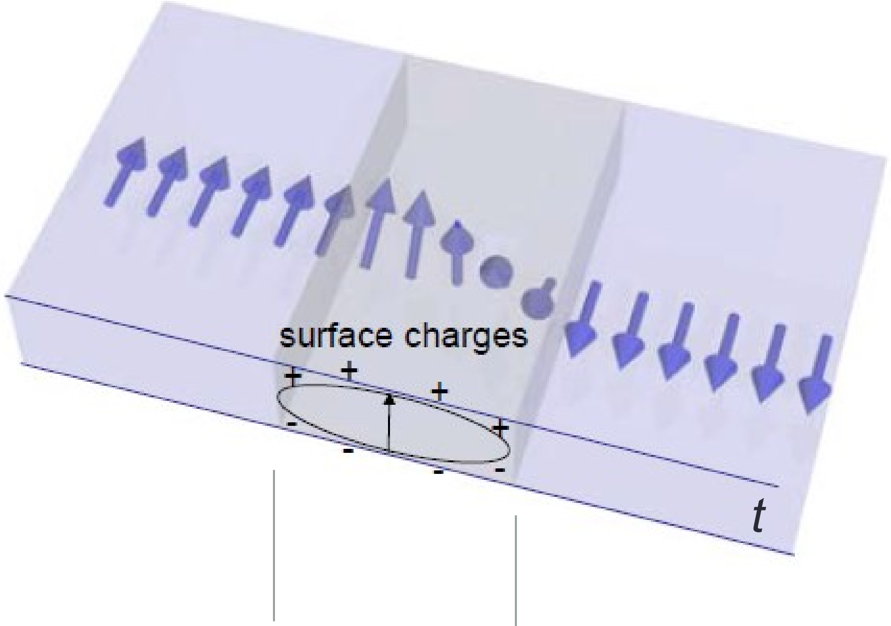


B. Pfau et al., Nature Communications, Vol. 3, 11; DOI:doi:10.1038/ncomms2108 (2012)
L.Müller et al., Rev. Sci. Instrum. 84, 013906 (2013)

Ferromagnetism in a nutshell – Domains and Walls

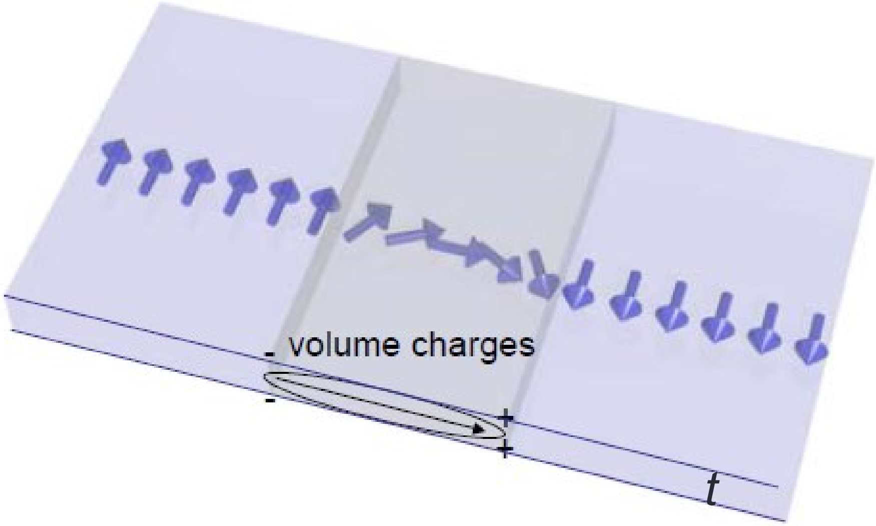
- Néel and Bloch domain walls for in-plane magnetized systems

Bloch wall



Domain wall width $d_w < t$

Néel wall

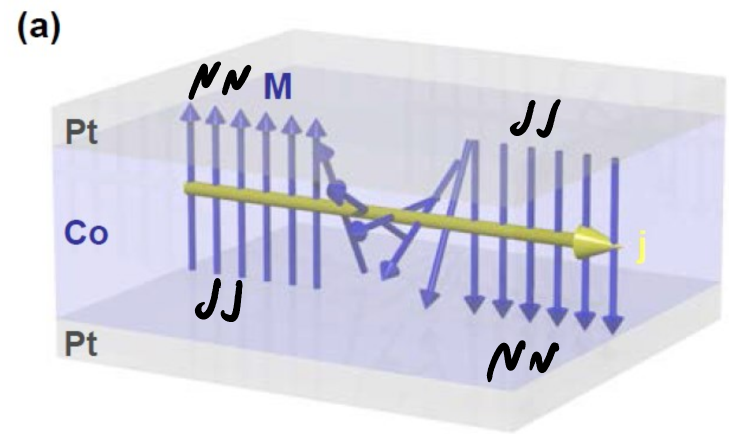


Domain wall width $d_w > t$

Ferromagnetism in a nutshell – Domains and Walls

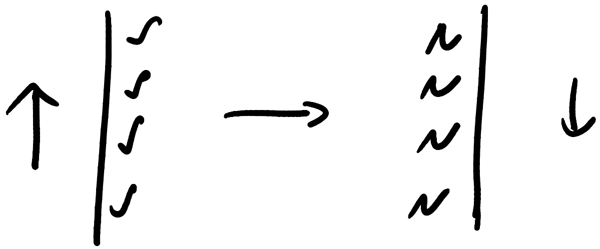
- Néel and Bloch domain walls for films with perpendicular anisotropy

Bloch wall



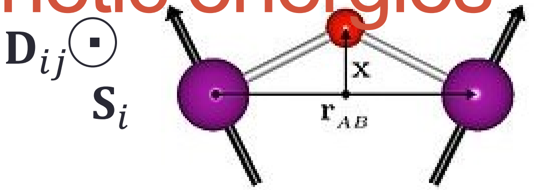
- No magnetic charges in wall!

Néel wall



- exhibits volume charges (unfavorable due to magnetostatic energy)
  but
- Néel wall favored by Dzyaloshinskii-Moriya interaction (considered since 2013!)

Ferromagnetism in a nutshell – Magnetic energies



DMI-Vector

- Dzyaloshinskii-Moriya interaction (DMI):
  Asymmetric exchange interaction

$$E_{\mathrm{DMI}} = \sum_{i \neq j} \mathbf{D}_{ij}\ (\mathbf{S}_i \times \mathbf{S}_j)$$

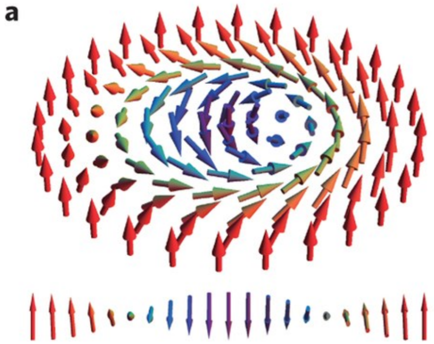
➔ Minimization of total energy yields to formation of chiral structures = ‘skyrmions’

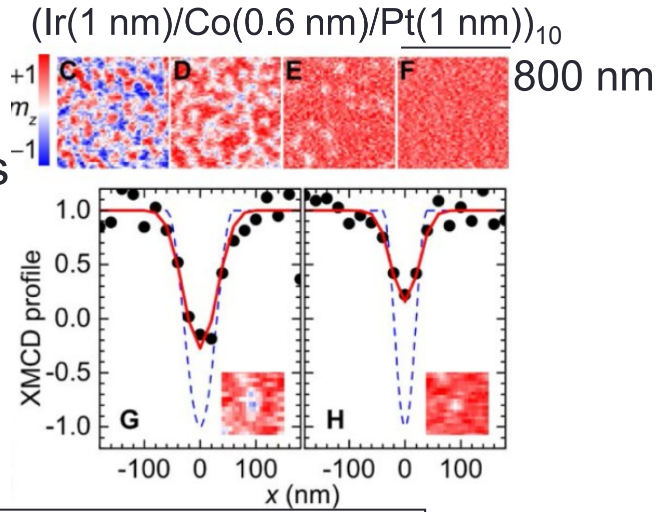
- **Asymmetric magnetic multilayers like Pt/Co/Ir**

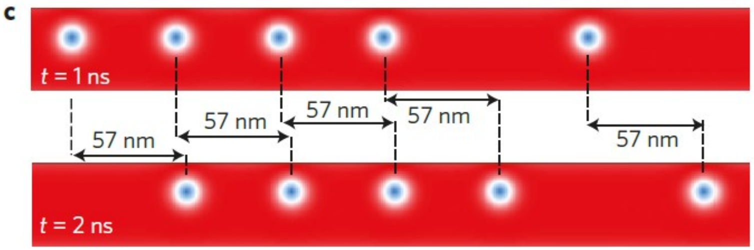
  Different sign of $\mathbf{D}_{ij}$ for Co/Pt and Co/Ir interface

  ➔ additive, large effective DMI

- Future Skyrmion-based memory & data storage devices







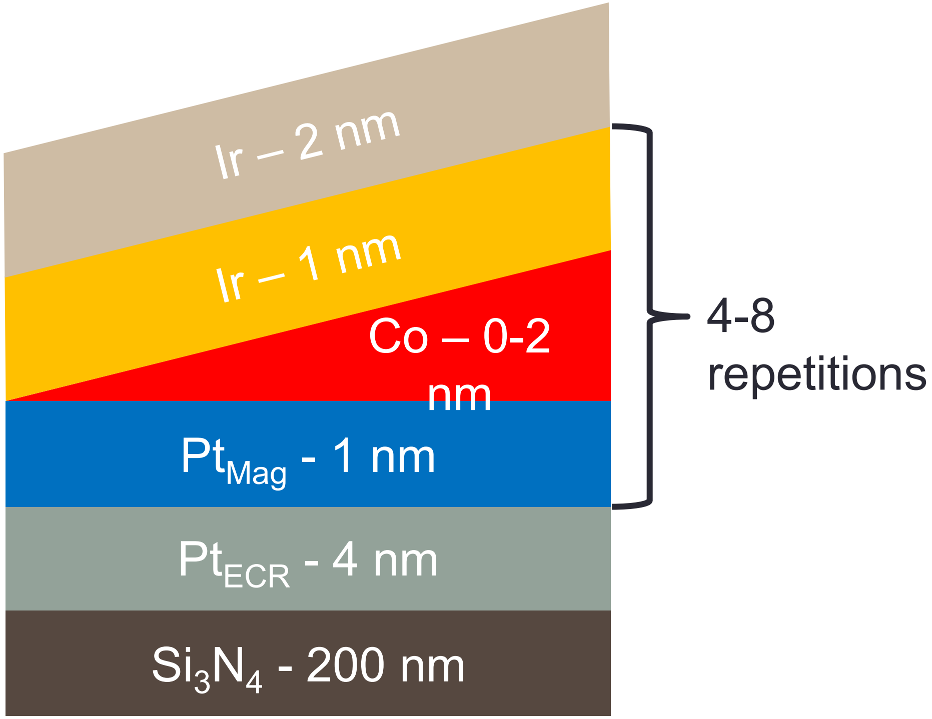
J. Sampaio, A. Fert et al., Nat. Nanotech. 8, 839 (2013)

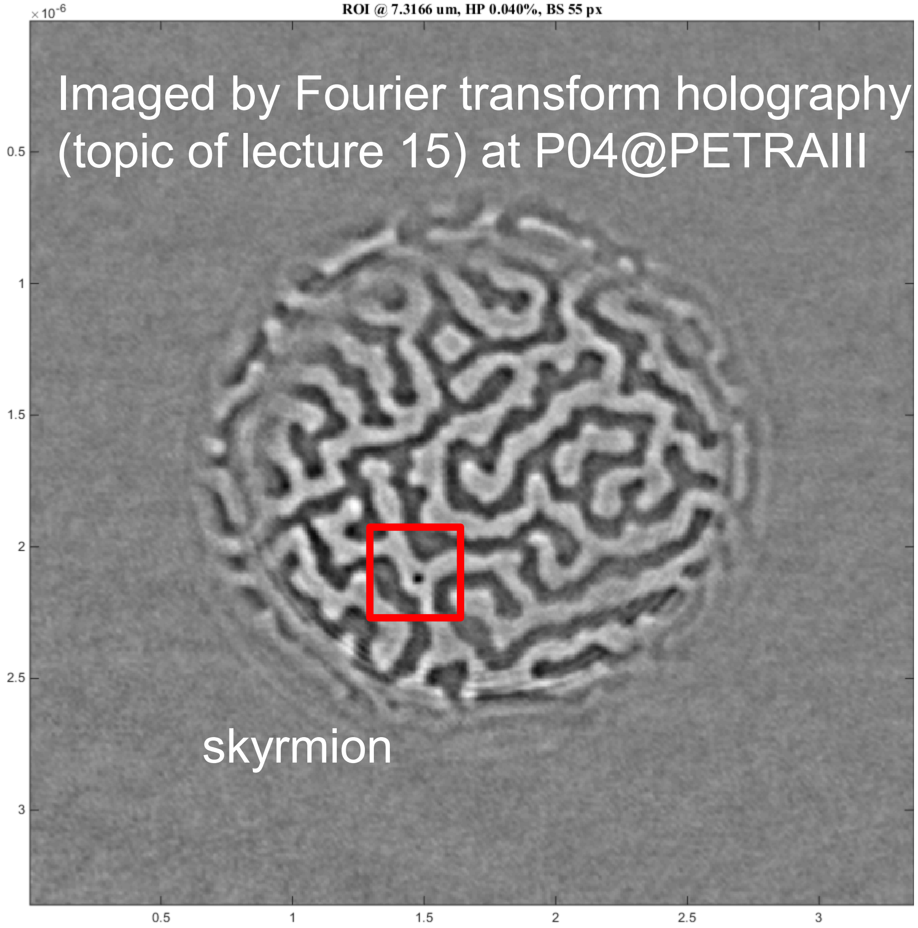
C. Moreau-Luchaire, A. Fert et al., arXiv:1502.07853v1 (2015)

Ferromagnetism in a nutshell – Domains and Walls

- What we are doing

Pt/Co/Ir Multilayers

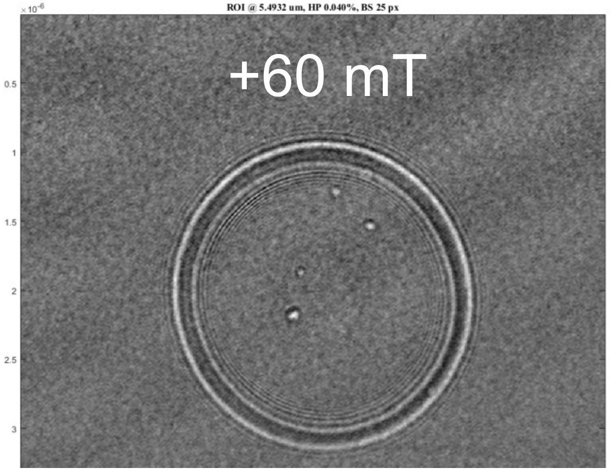


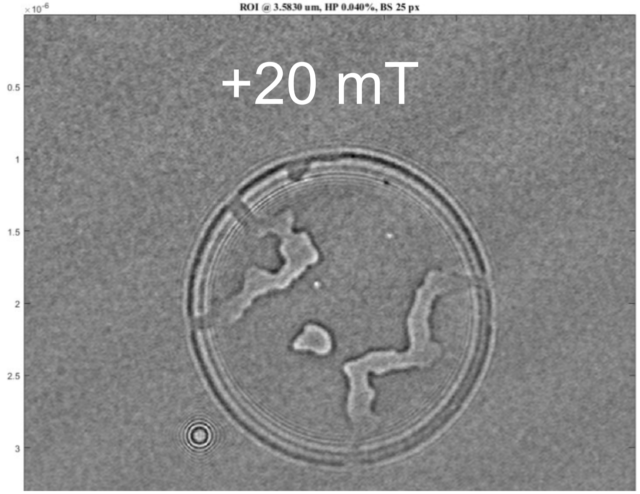


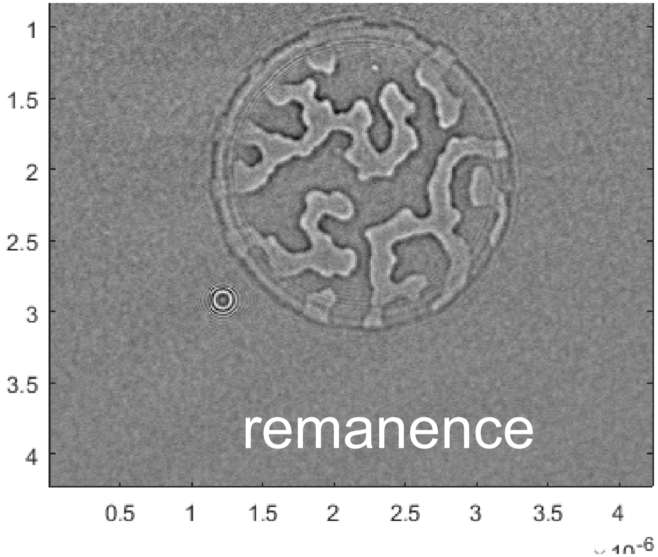
Imaged by Fourier transform holography (topic of lecture 15) at P04@PETRAIII

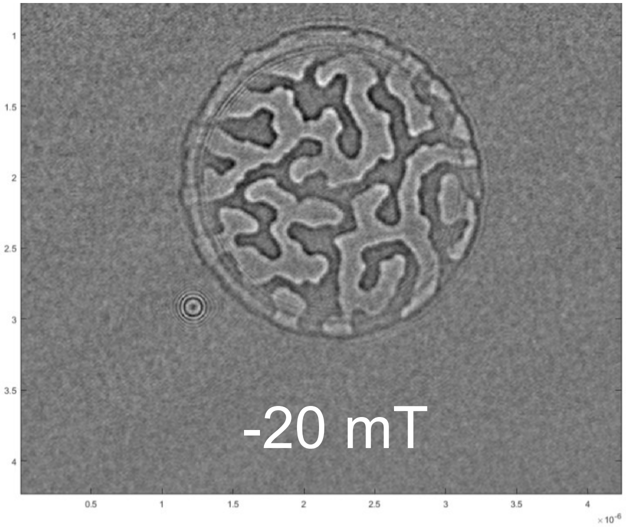
Ferromagnetism in a nutshell – Domains and Walls

- Out-of-plane field sweep



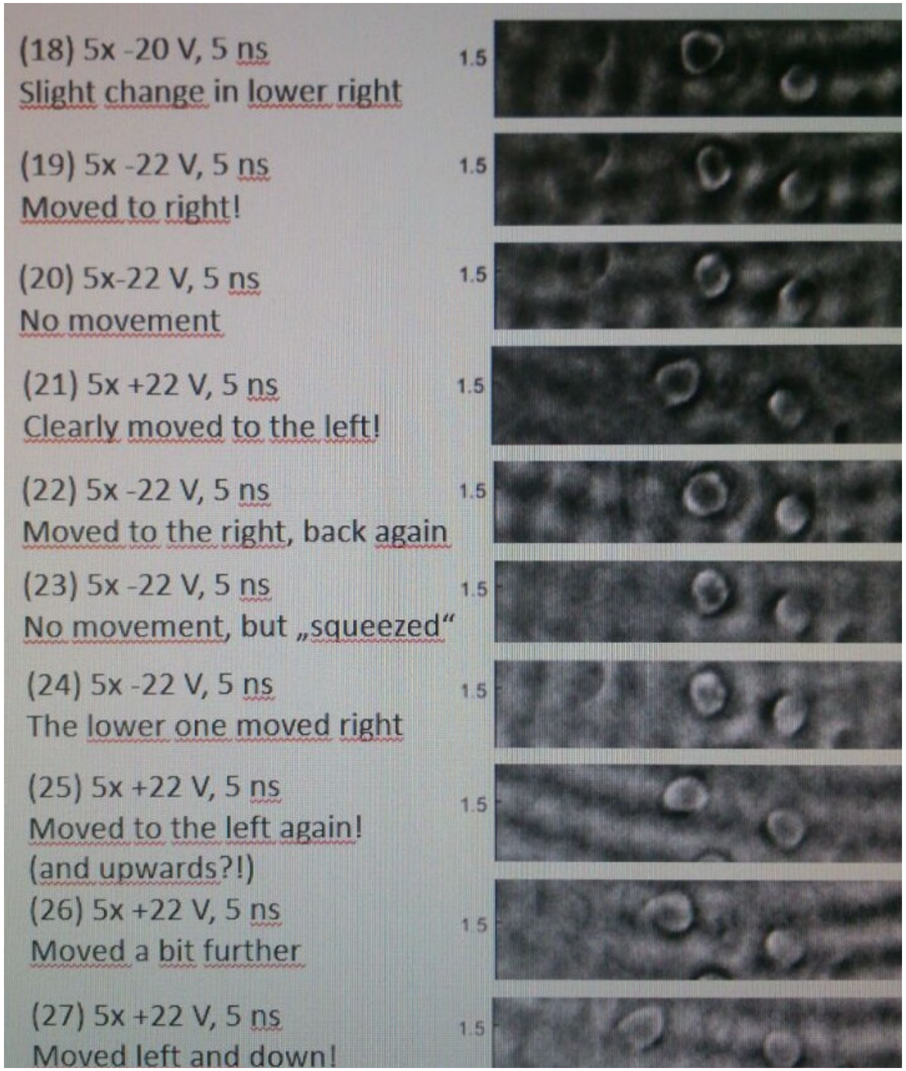






Ferromagnetism in a nutshell – Domains and Walls

- Current induced Skyrmion motion



(18) 5x -20 V, 5 ns
Slight change in lower right

(19) 5x -22 V, 5 ns
Moved to right!

(20) 5x-22 V, 5 ns
No movement

(21) 5x +22 V, 5 ns
Clearly moved to the left!

(22) 5x -22 V, 5 ns
Moved to the right, back again

(23) 5x -22 V, 5 ns
No movement, but „squeezed“

(24) 5x -22 V, 5 ns
The lower one moved right

(25) 5x +22 V, 5 ns
Moved to the left again!
(and upwards?!)

(26) 5x +22 V, 5 ns
Moved a bit further

(27) 5x +22 V, 5 ns
Moved left and down!